



The Center for

BRIGHTBEAMS

A National Science Foundation Science & Technology Center



First principles study on the impact of grain boundaries on Nb₃Sn

9:30am Tuesday 10 November 2020

Virtual International Workshop on Nb₃Sn SRF Science, Technology & Applications

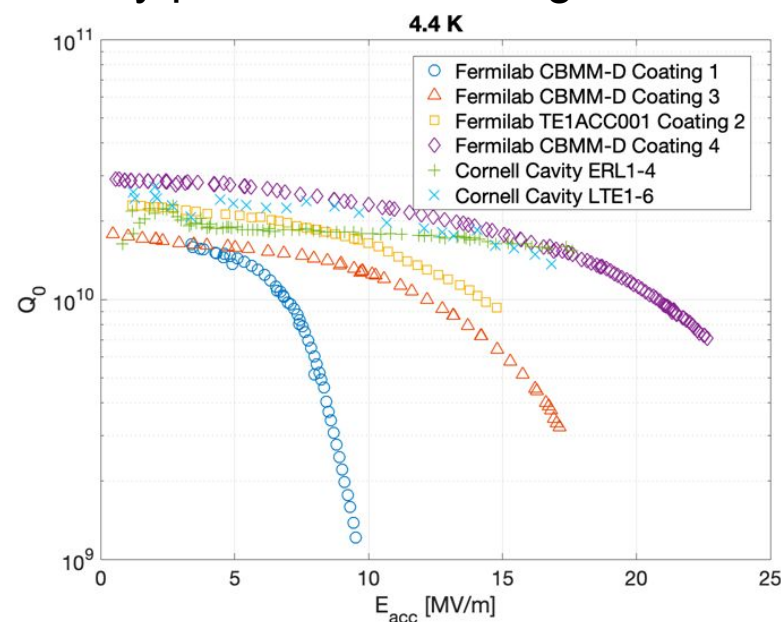
Michelle Kelley, Nathan Sitaraman and Tomás Arias (Cornell University)



Nb₃Sn grain boundaries & cavity performance



Nb₃Sn SRF cavity studies¹ indicate correlation between cavity performance and grain boundary (GB) composition.



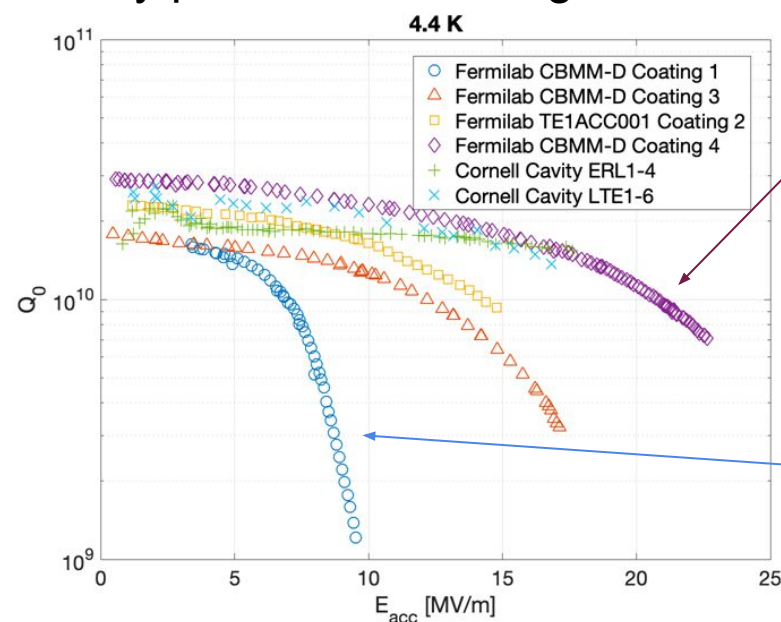
[1] Lee et al. (2020) Acta Mat. 188, 1505



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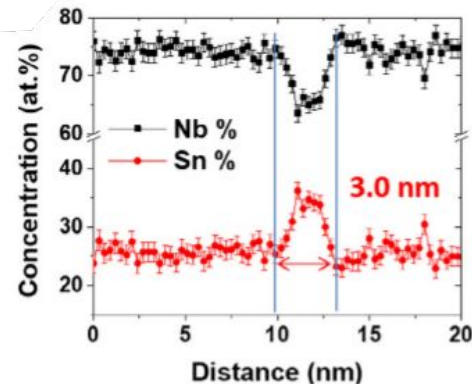


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Stable Q in cavities with
“clean” GBs

Degraded Q in cavities
with **excess tin at GBs!**

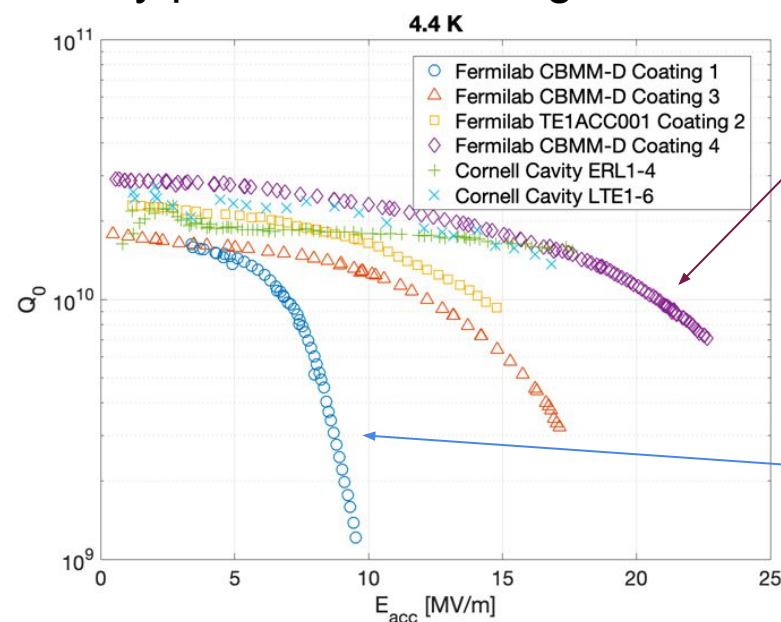




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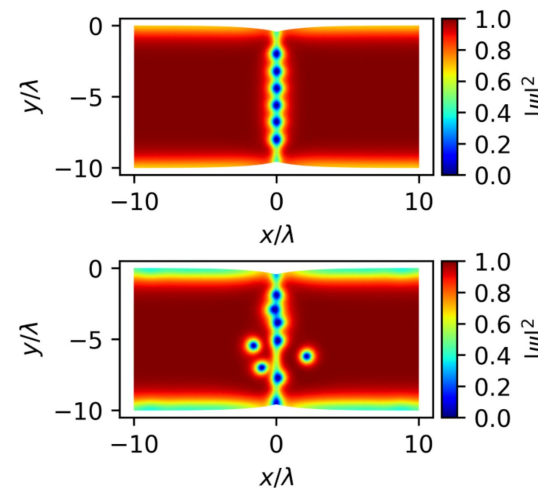
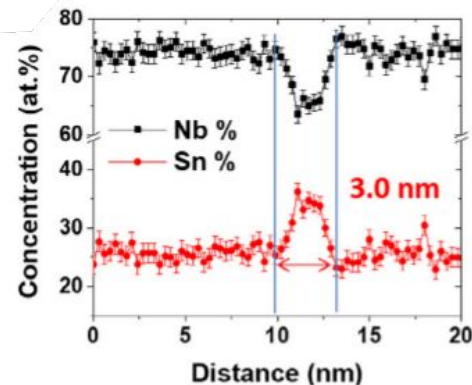


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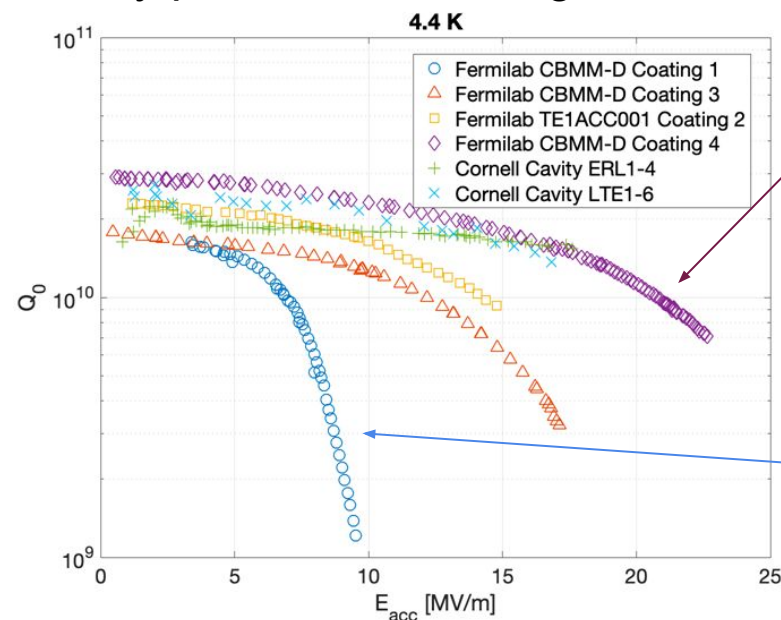
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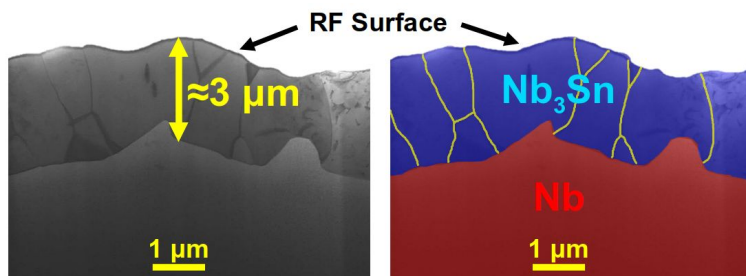
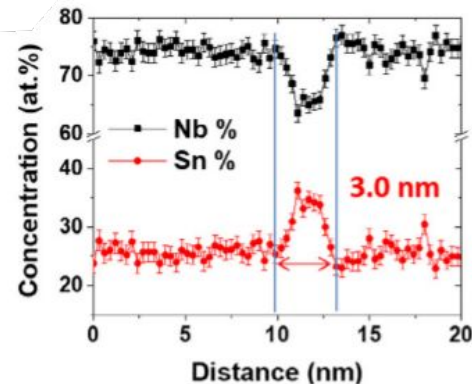


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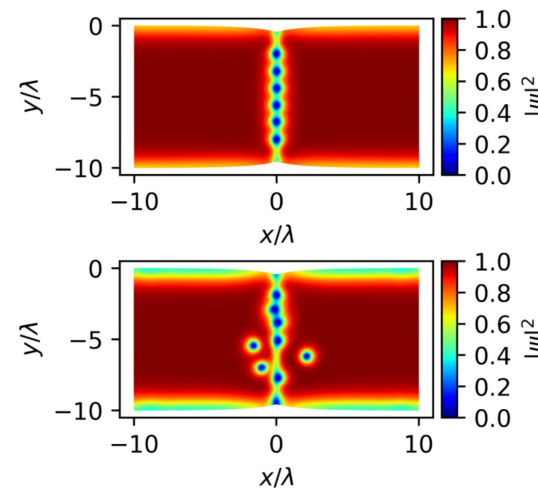


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	T _c (K)	ξ (nm)
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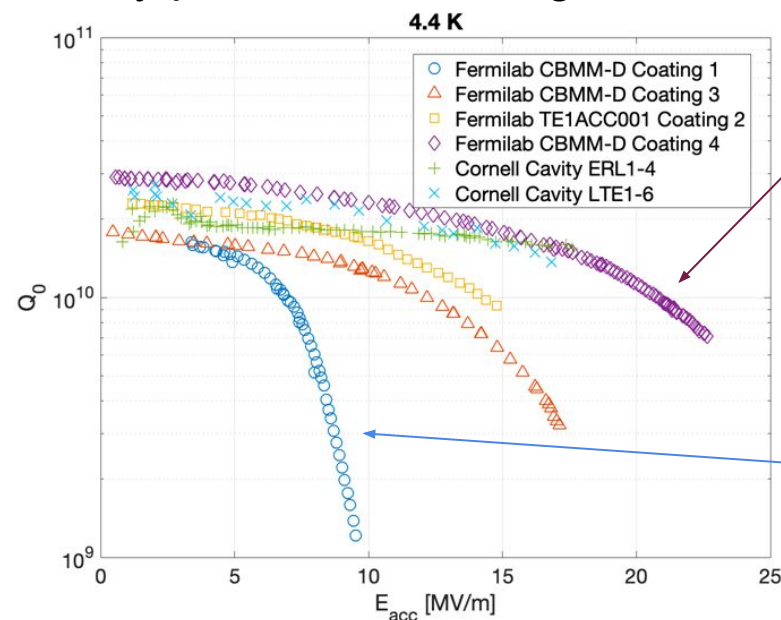
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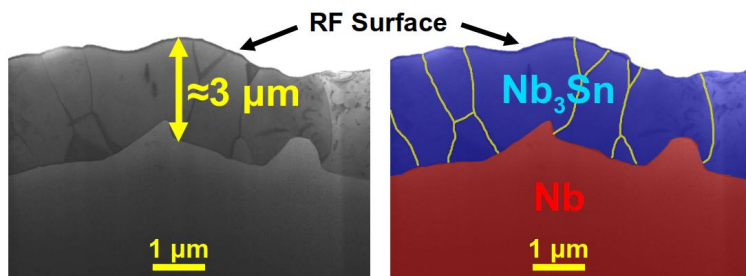
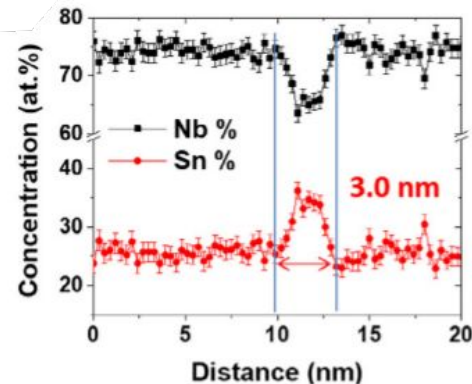


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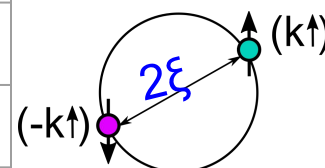


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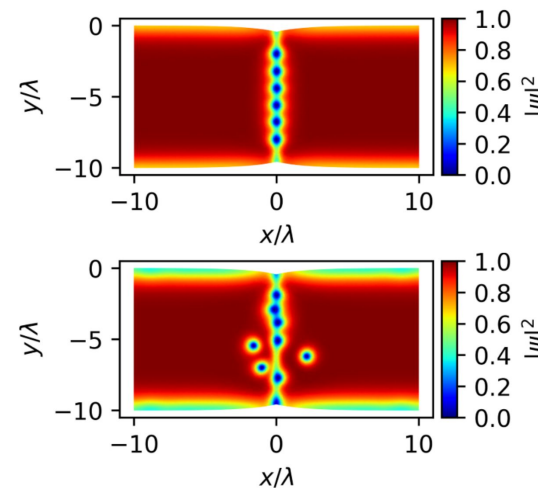
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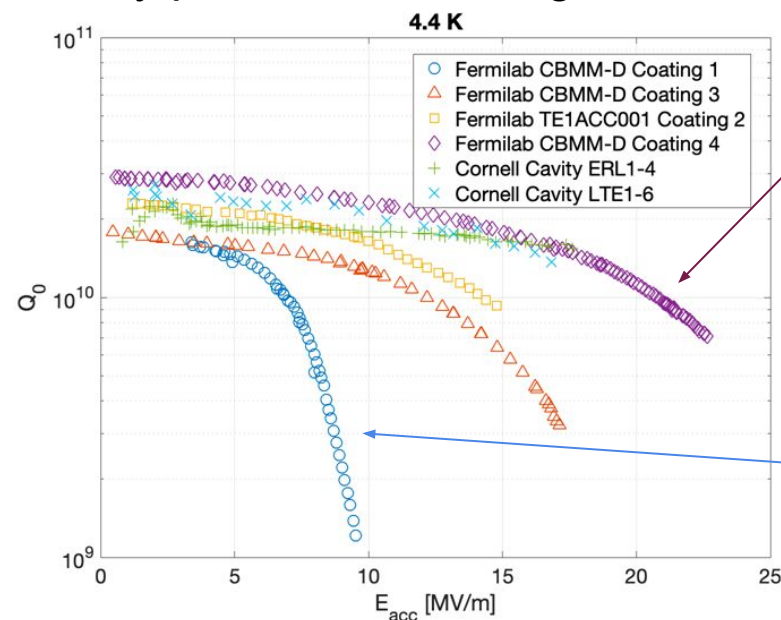
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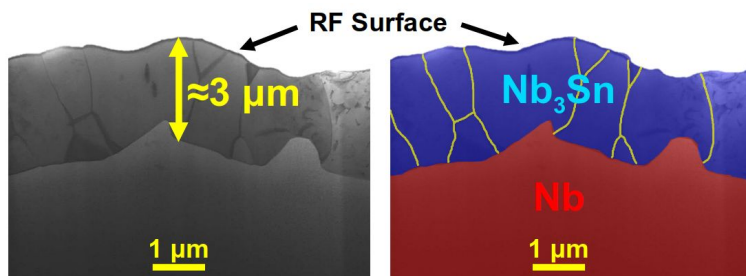
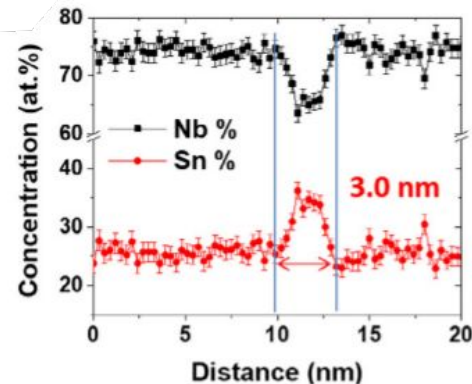


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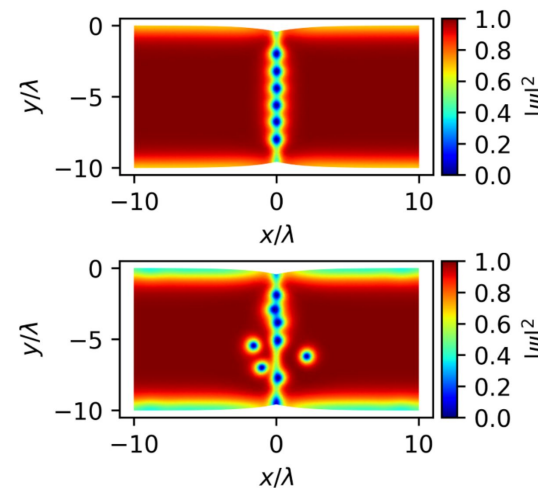
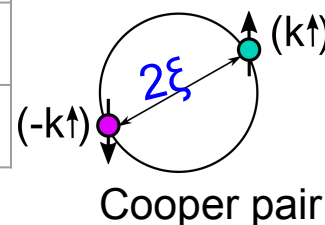


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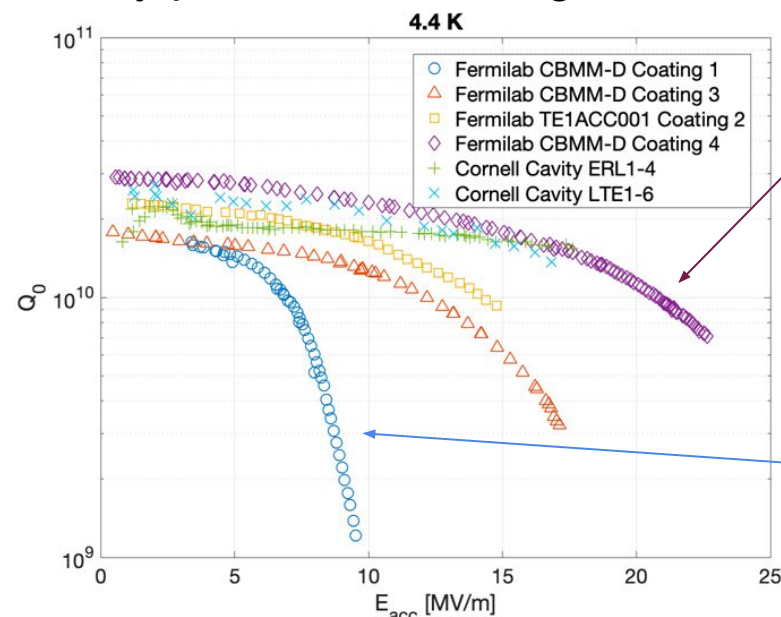
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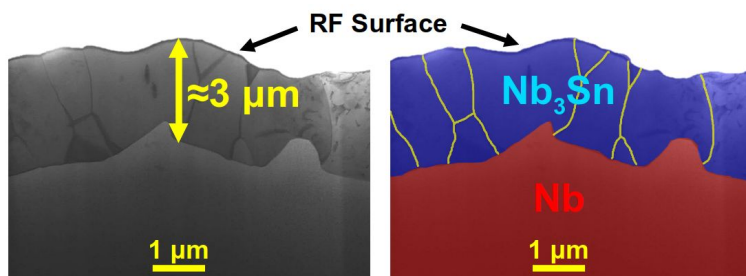
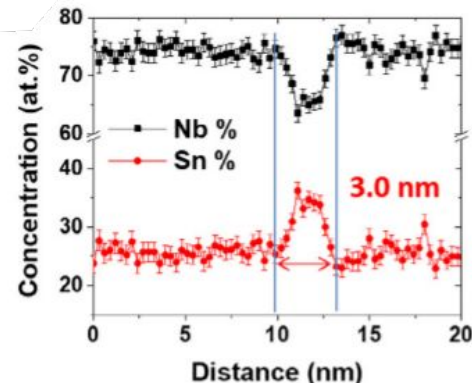


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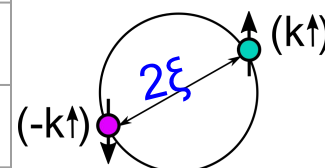


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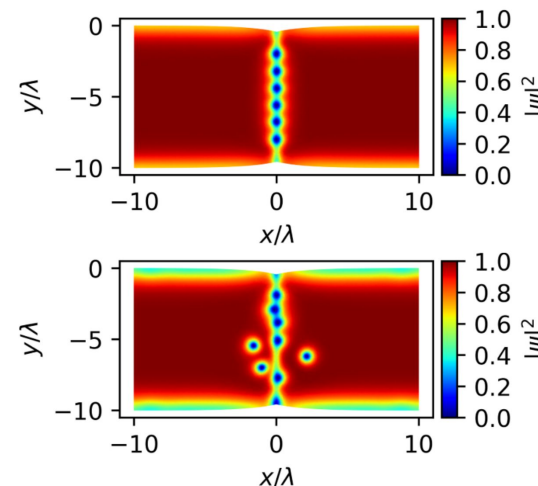
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- act as the primary flux pinning centers
- often exhibit enhanced concentrations of Sn or ternary elements like Cu and Ti

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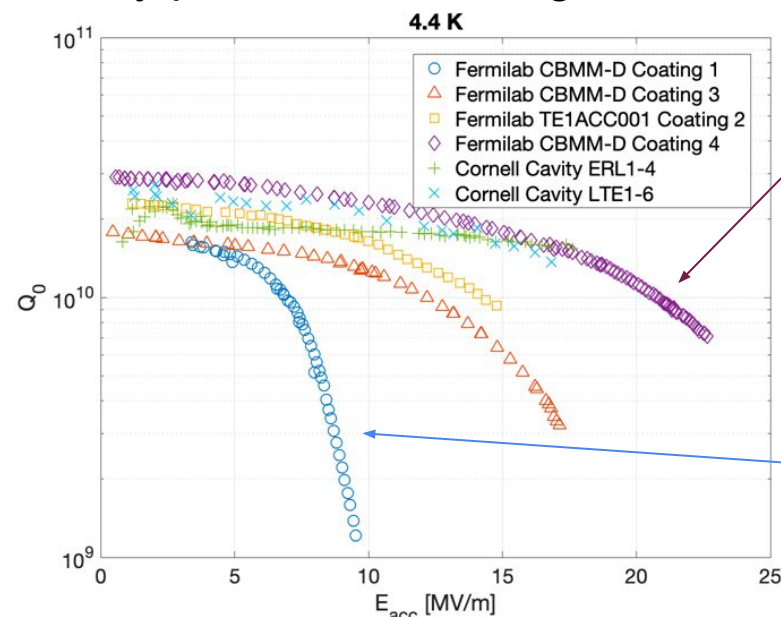
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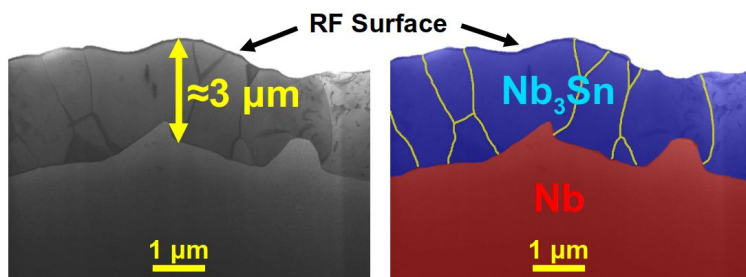
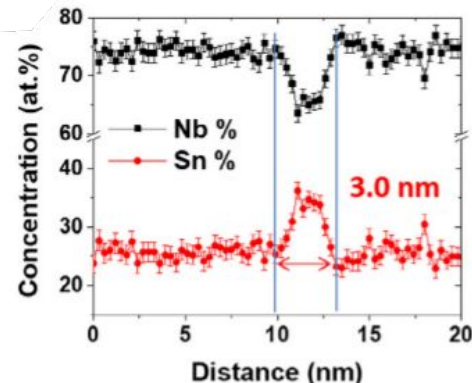


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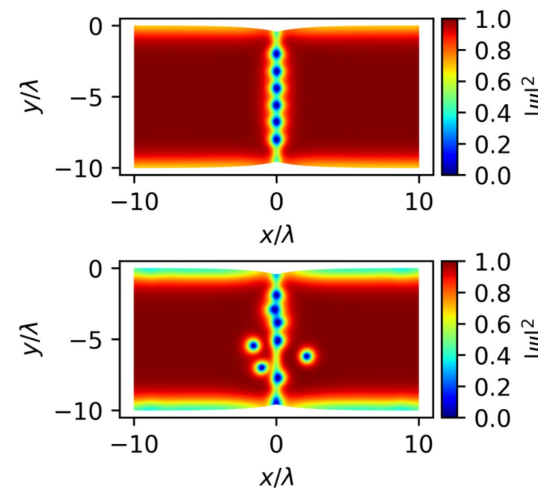
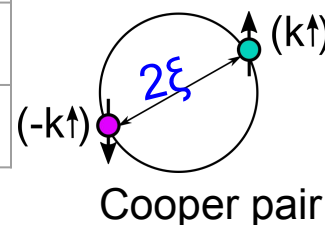


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- often exhibit enhanced concentrations of Sn or ternary elements like Cu and Ti
- *have surprisingly not been studied in detail using density-functional theory (DFT)*

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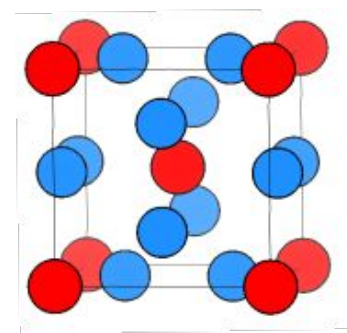
Superconducting Nb₃Sn (undefected material)



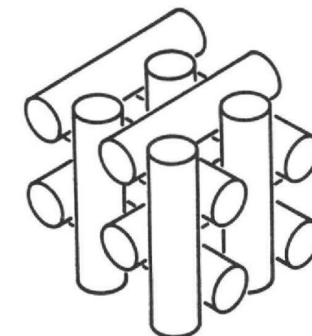
A15 Superconductors:

highest T_c from 1954-1986

Nb₃Sn: T_c ≈ 18 K



A15 Unit Cell



Inorg. Chem. 1990, 29, 2164-2190



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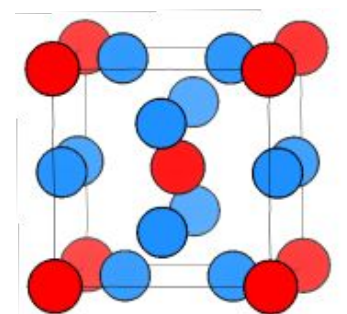
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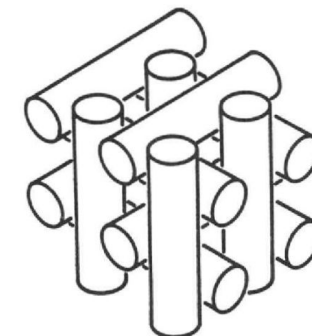
Debye cut-off energy

Electron-phonon
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Density of states at
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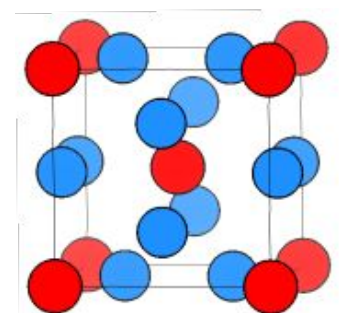
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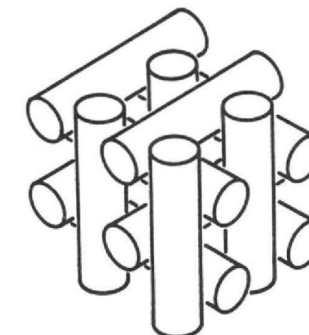
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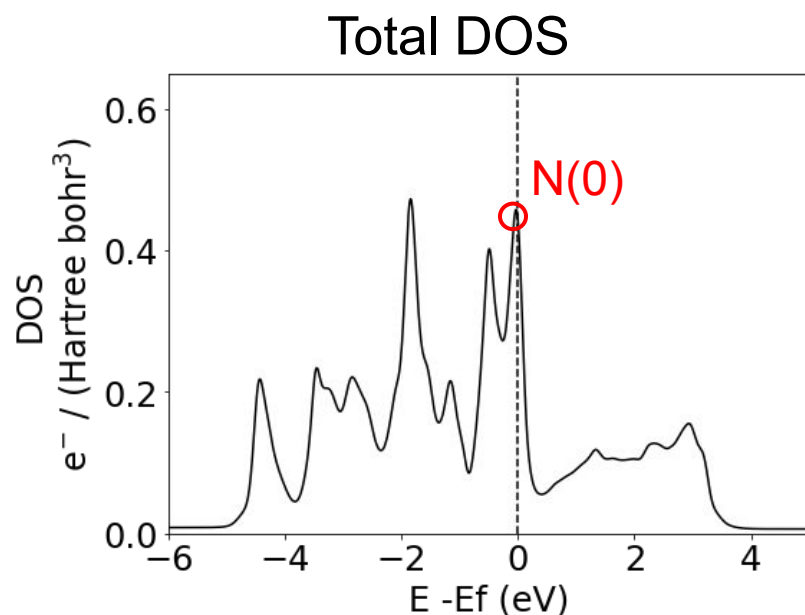
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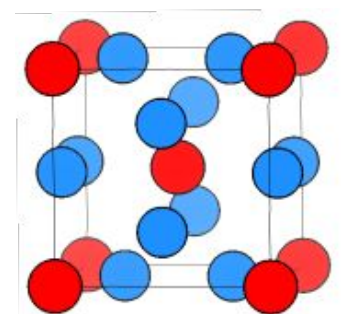
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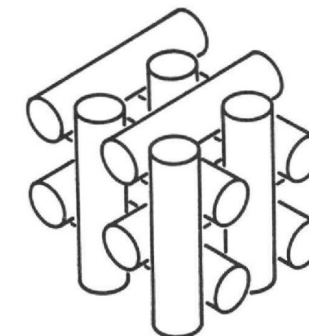
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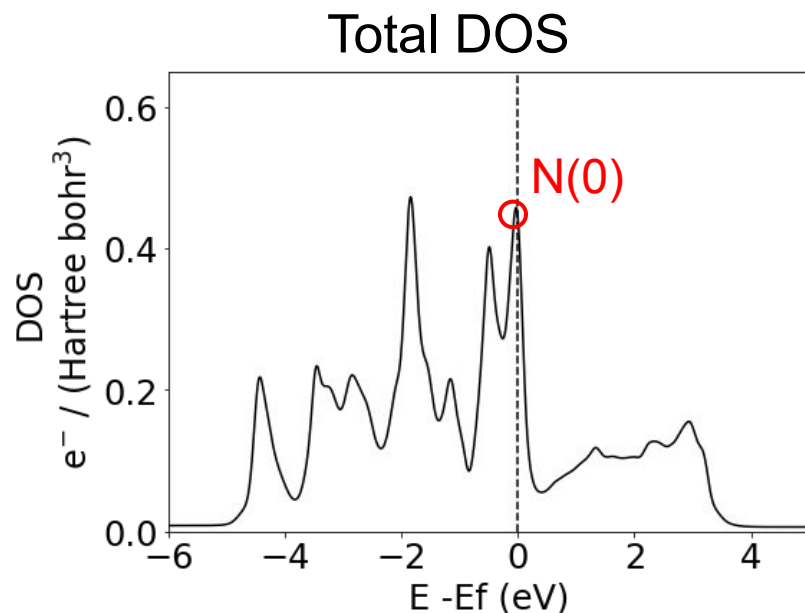
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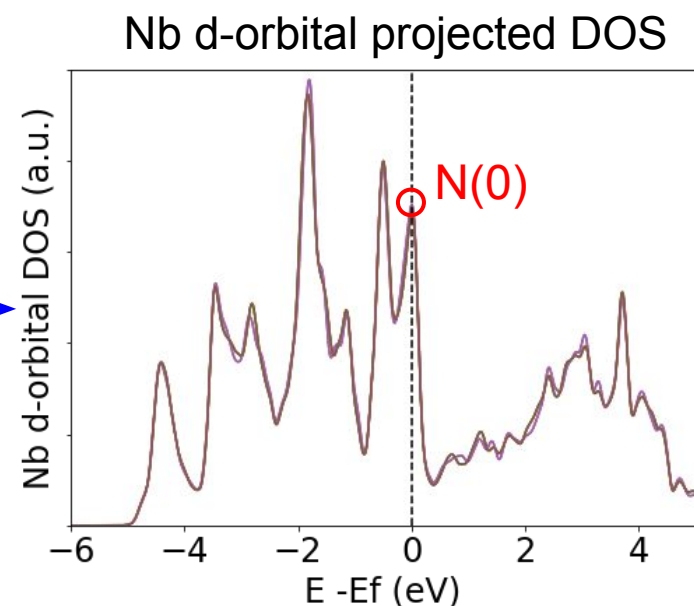


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Total DOS projected onto
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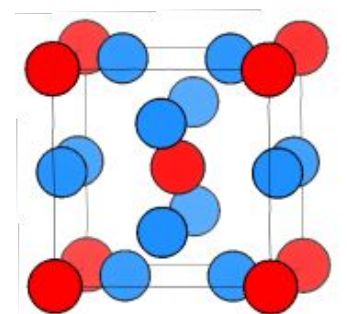
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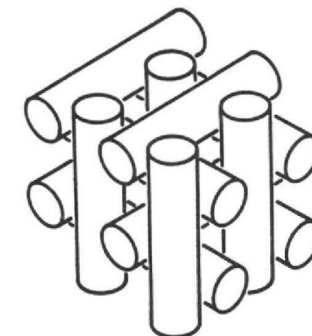
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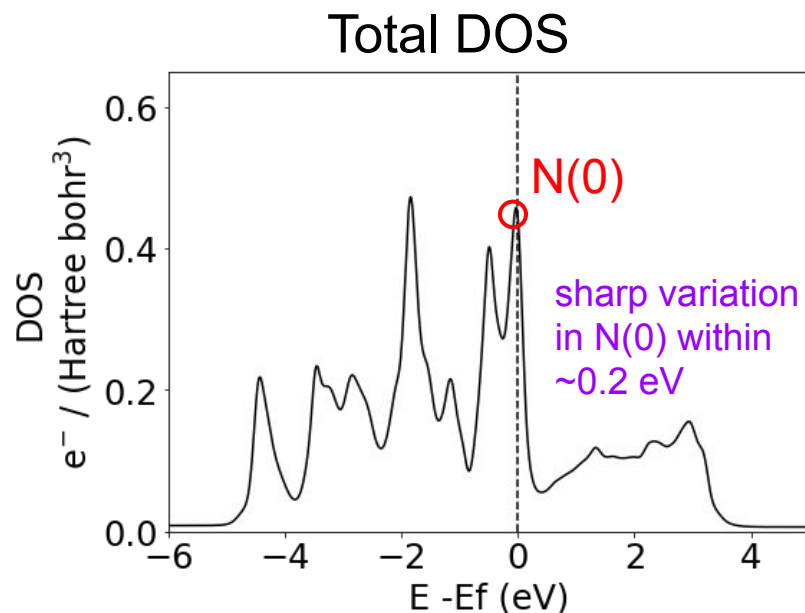
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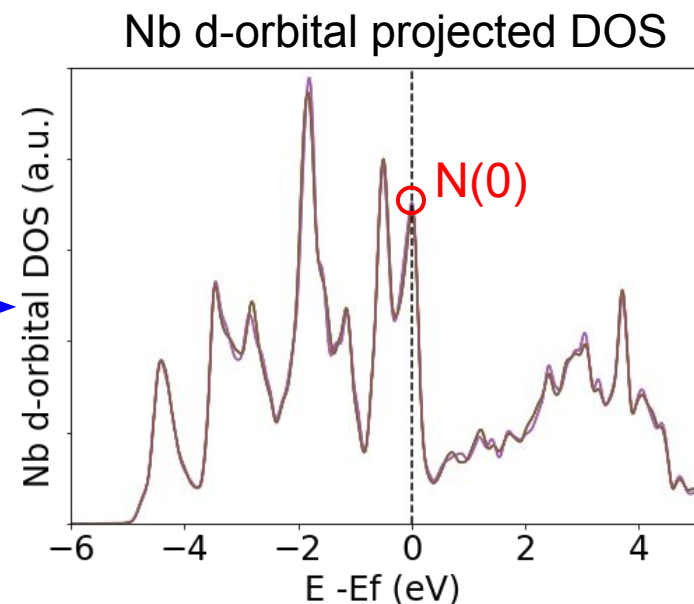


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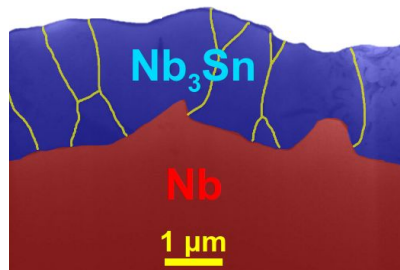
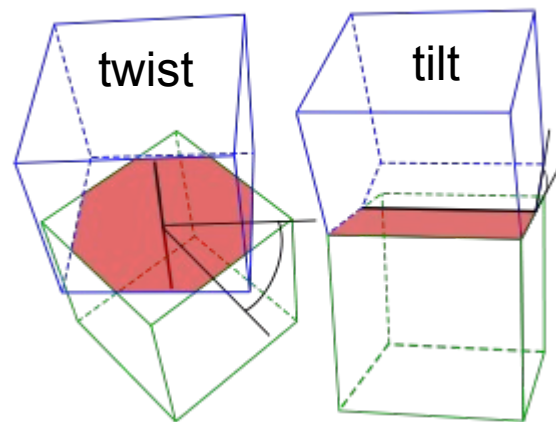
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Total DOS projected onto
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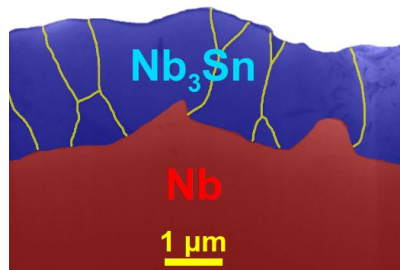
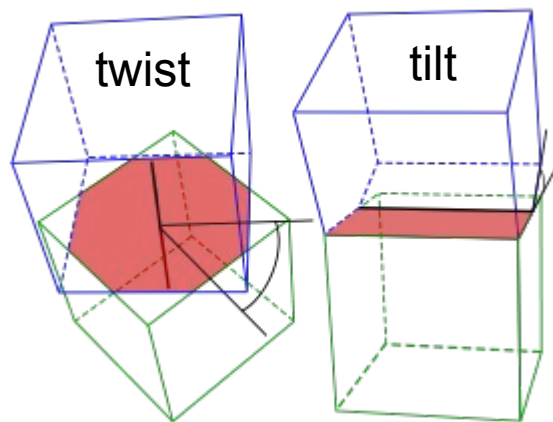


Grain boundaries in Nb_3Sn calculated with DFT

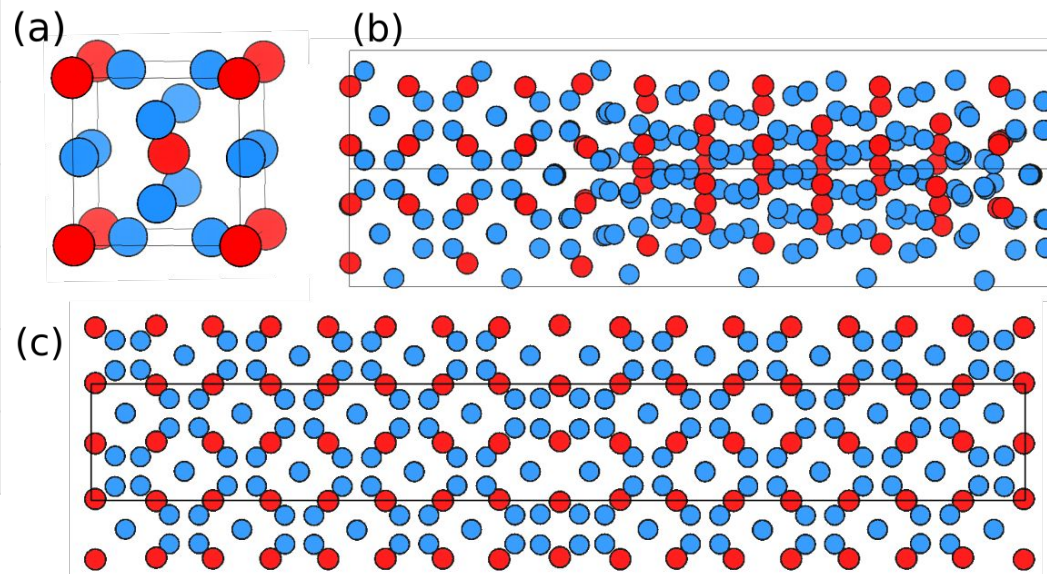




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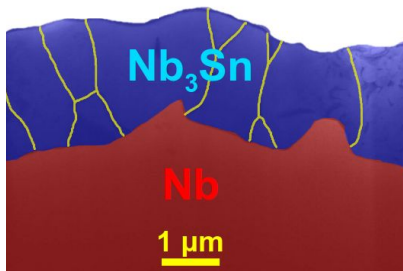
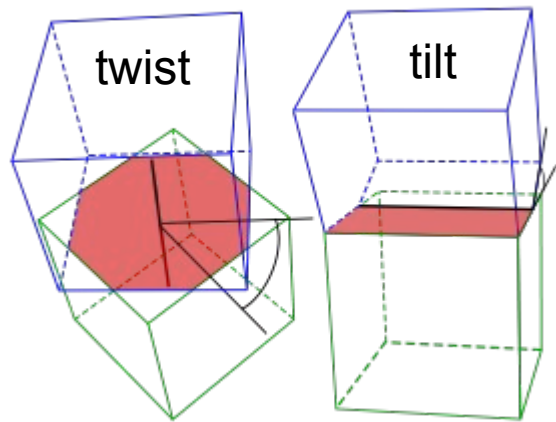


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(110)-tilt (c)	128	2.99	1455
(112)-tilt	192	2.59	840
(100)-twist	78	2.64	650
(110)-twist (b)	288	2.24	1300
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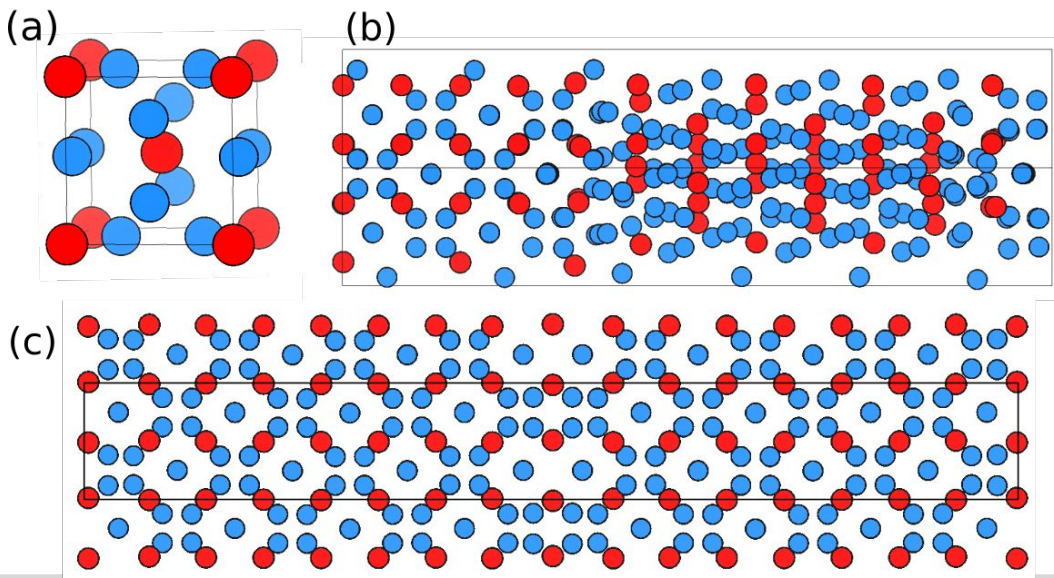
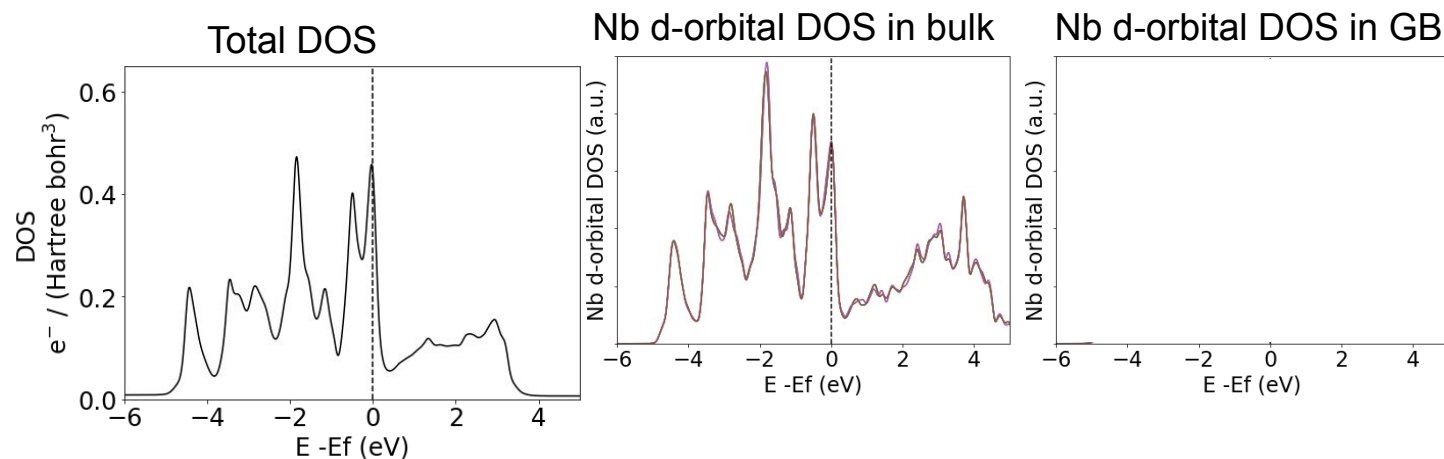




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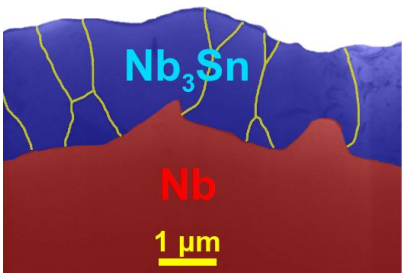
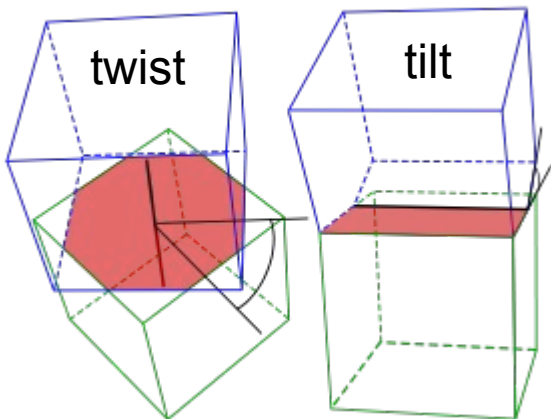
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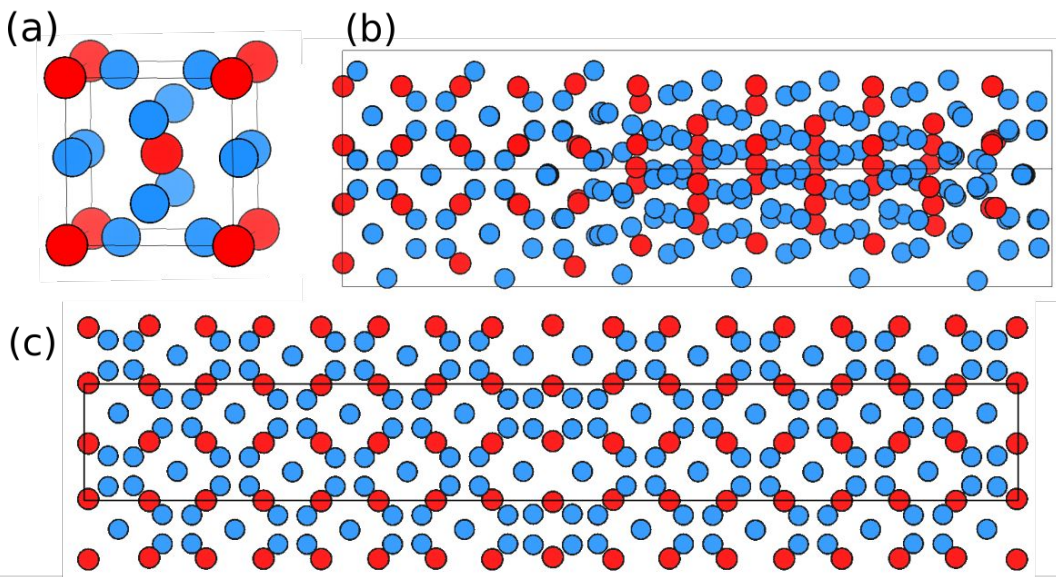
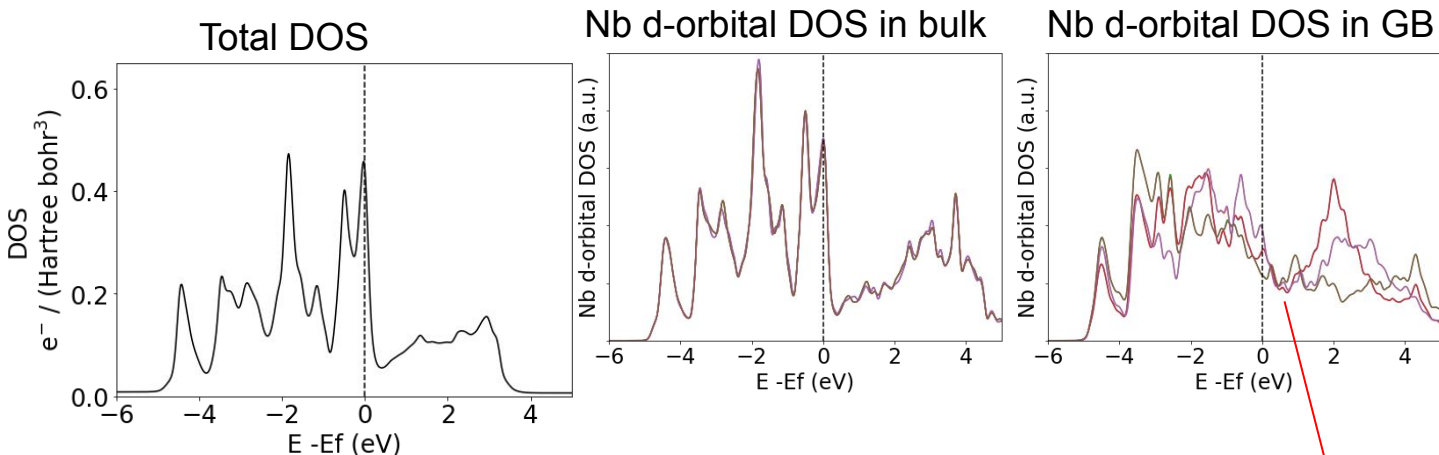
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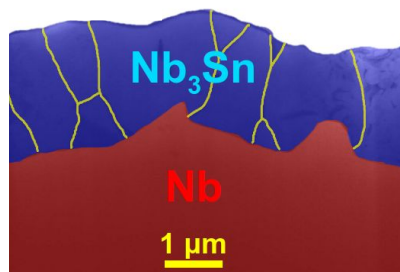
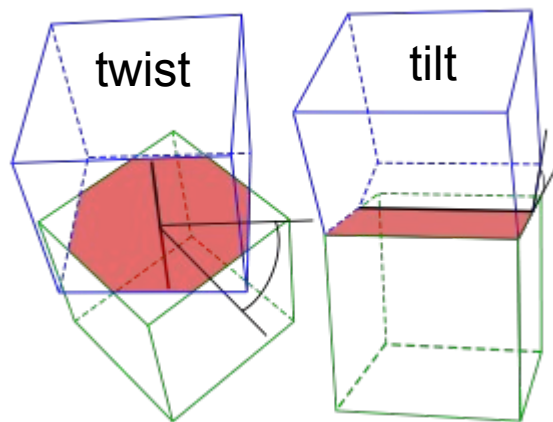


GBs reduce **N(0)** by > a factor of 2!

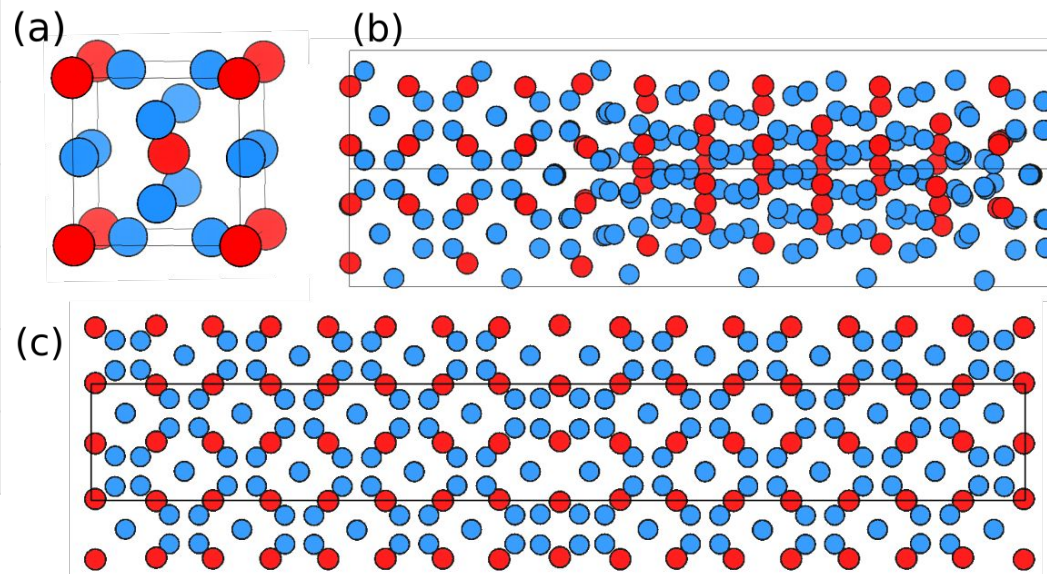
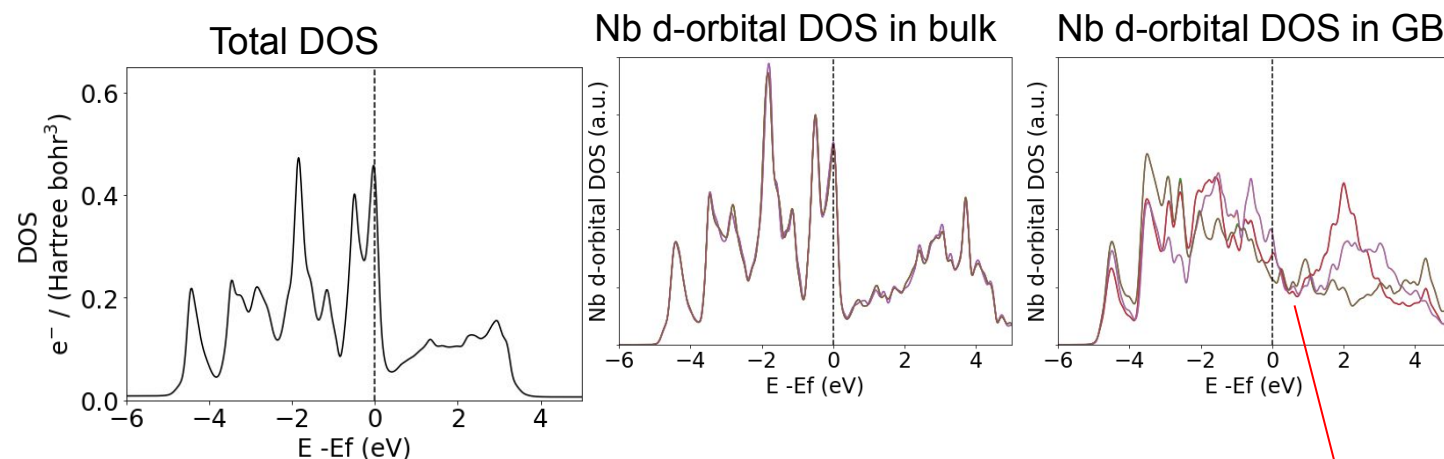
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Global reduction?
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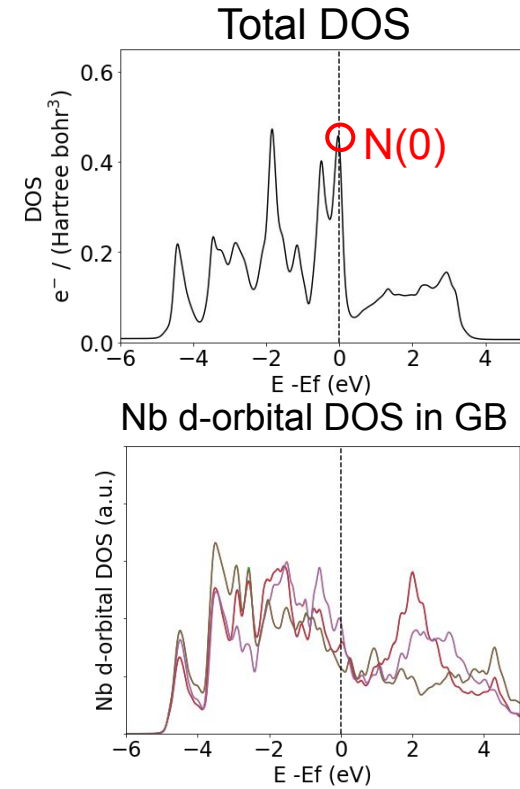
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Local DOS vs distance from a “clean” GB



Global reduction in $N(0)$ by $>$ a factor of 2: $T_c \approx 18 \text{ K} \rightarrow T_c < 10 \text{ K}$ $k_B T_c = 1.13 E_D \exp(-1/(N(0)V))$



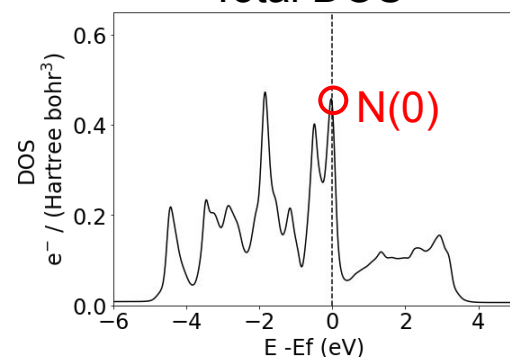


Local DOS vs distance from a “clean” GB

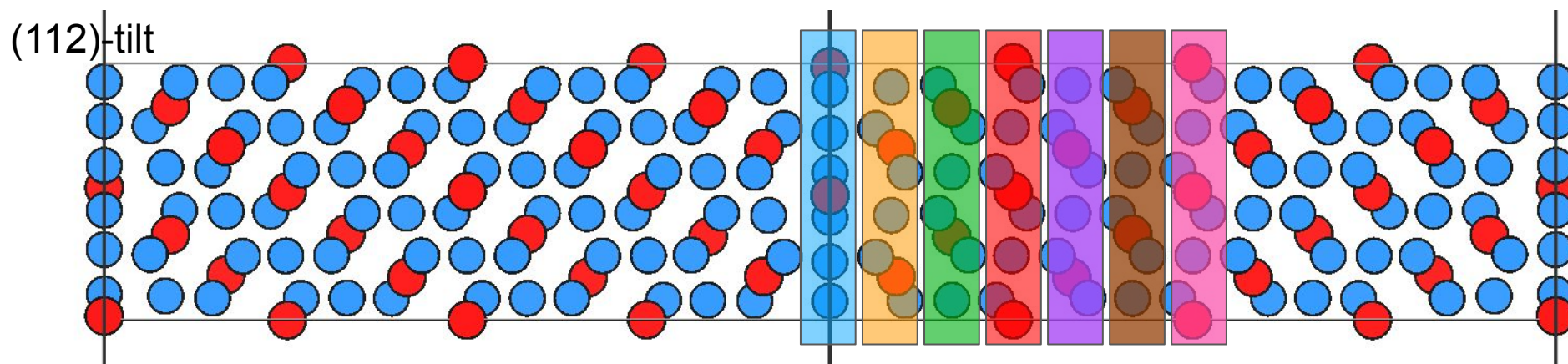
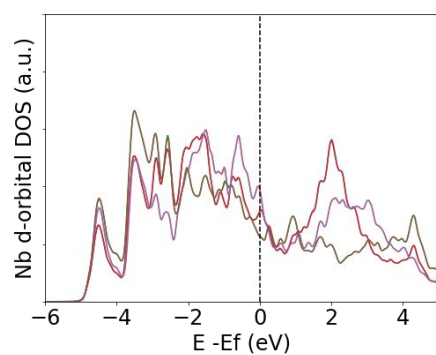


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Total DOS



Nb d-orbital DOS in GB



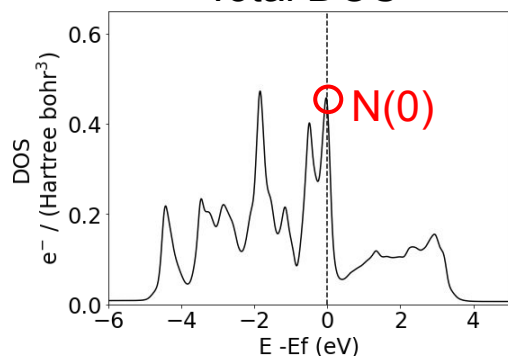


Local DOS vs distance from a “clean” GB

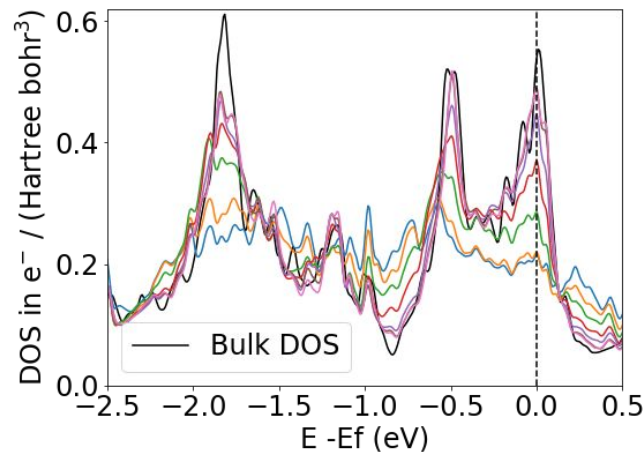
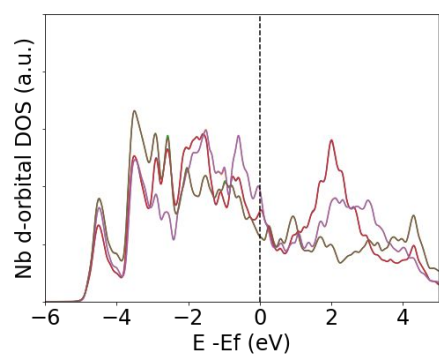


Global reduction in $N(0)$ by $>$ a factor of 2: $T_c \approx 18 \text{ K} \rightarrow T_c < 10 \text{ K}$ $k_B T_c = 1.13 E_D \exp(-1/(N(0)V))$

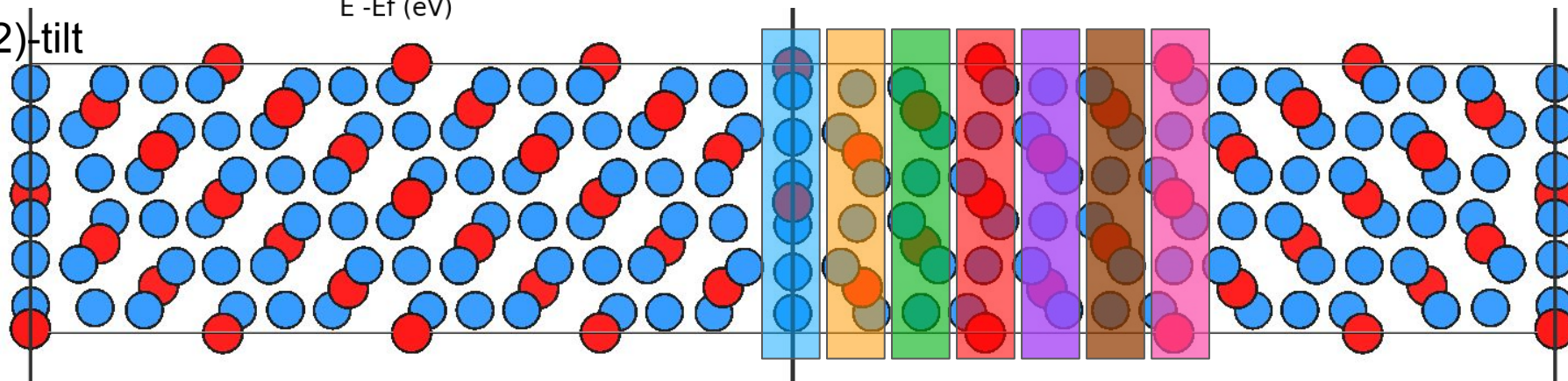
Total DOS



Nb d-orbital DOS in GB



(112)-tilt

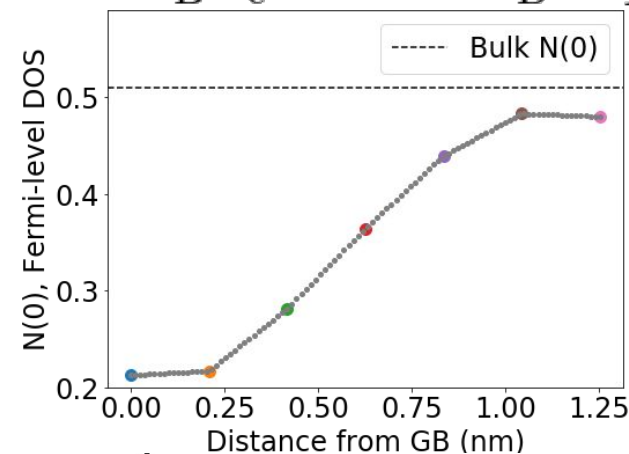
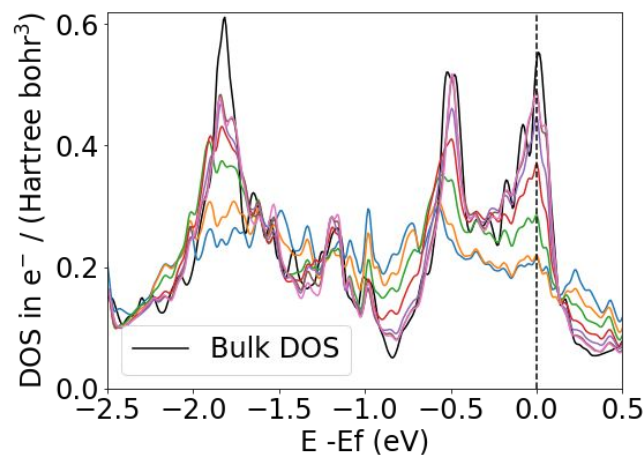
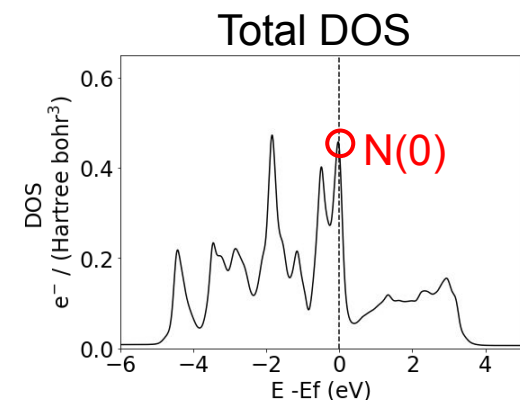




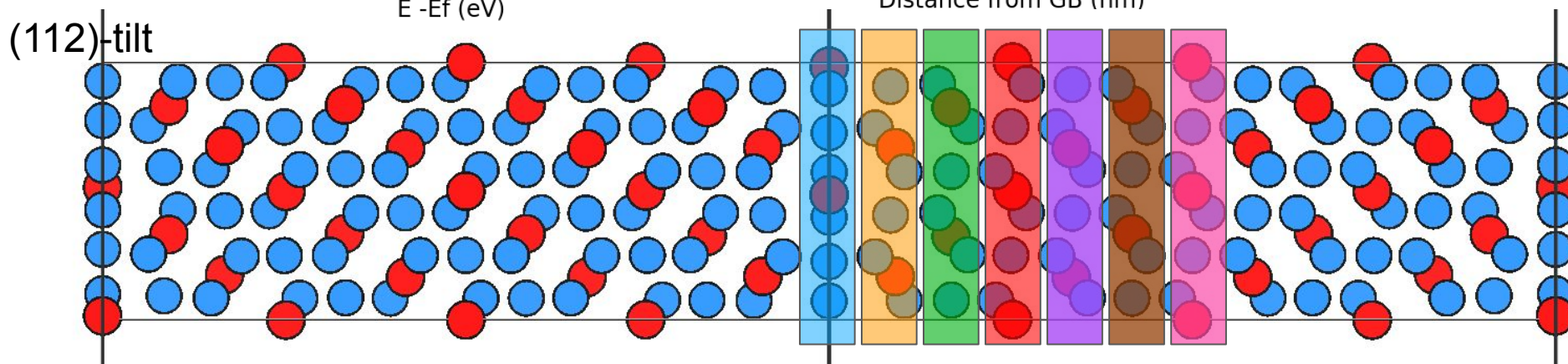
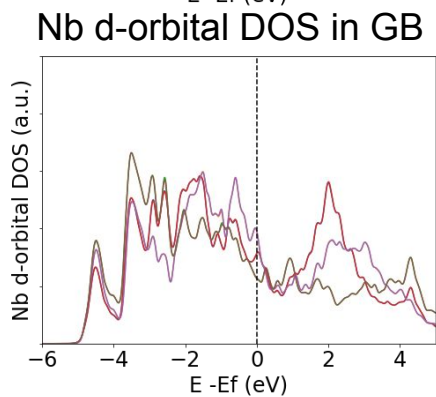
Local DOS vs distance from a “clean” GB



Global reduction in $N(0)$ by $>$ a factor of 2: $T_c \approx 18 \text{ K} \rightarrow T_c < 10 \text{ K}$ $k_B T_c = 1.13 E_D \exp(-1/(N(0)V))$



$N(0)$ relaxes back to bulk in $\sim 1\text{-}1.5 \text{ nm}$



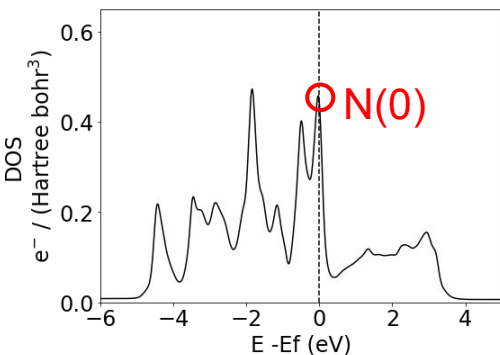


Local DOS vs distance from a “clean” GB

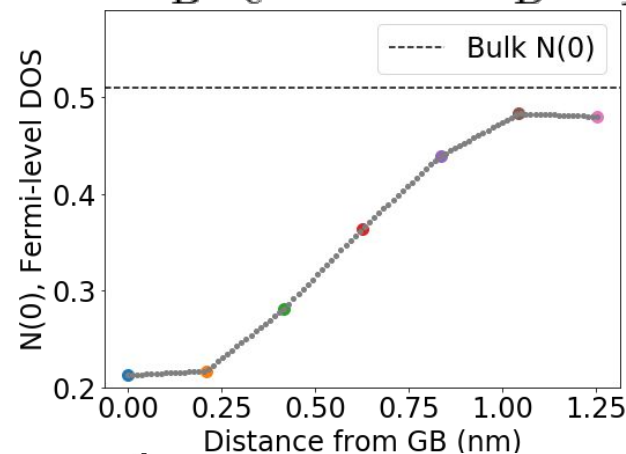
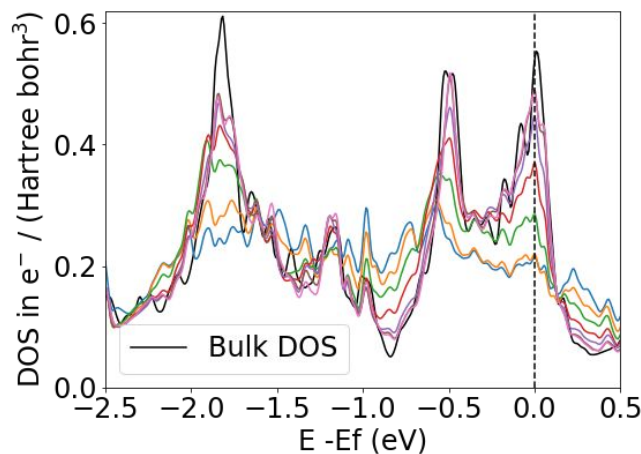
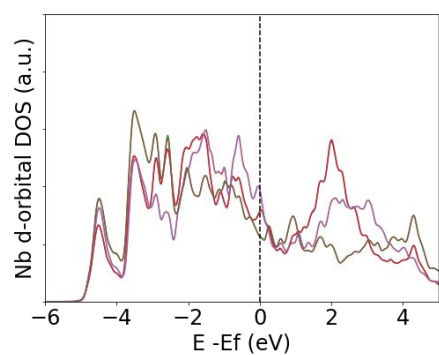


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Total DOS

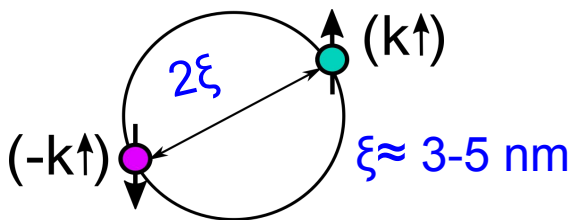
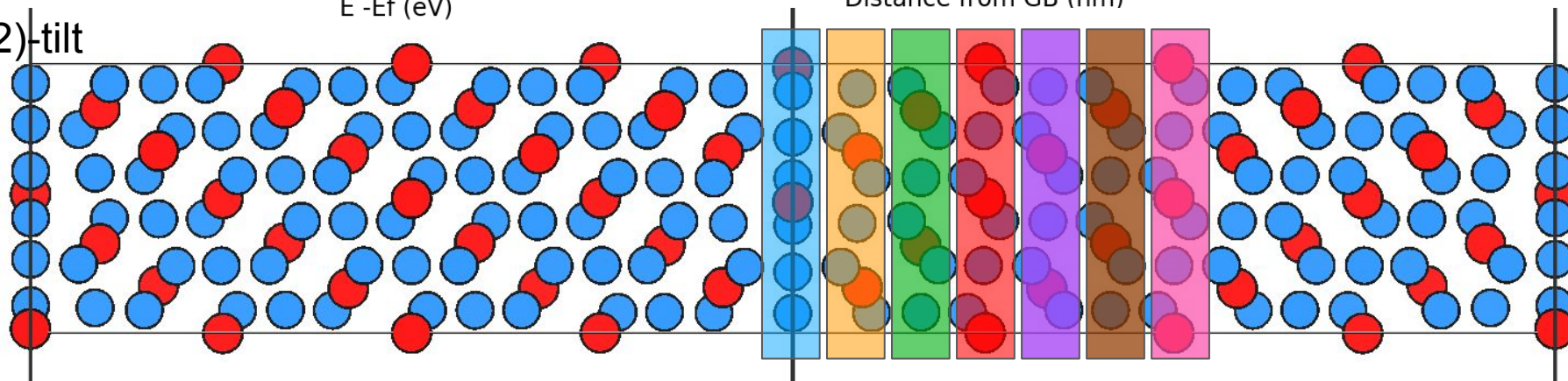


Nb d-orbital DOS in GB



$N(0)$ relaxes back to bulk in $\sim 1\text{-}1.5 \text{ nm}$

(112)-tilt

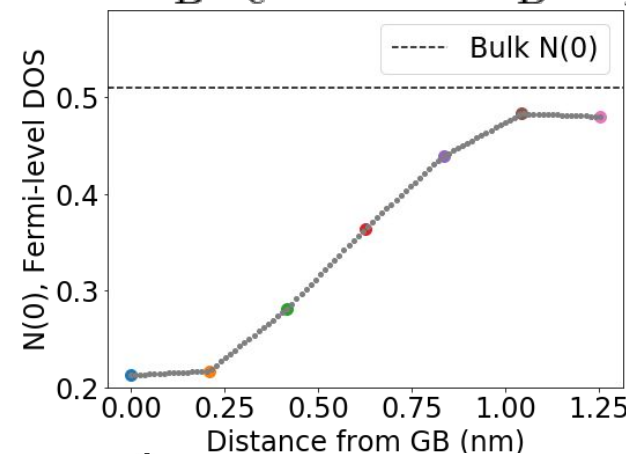
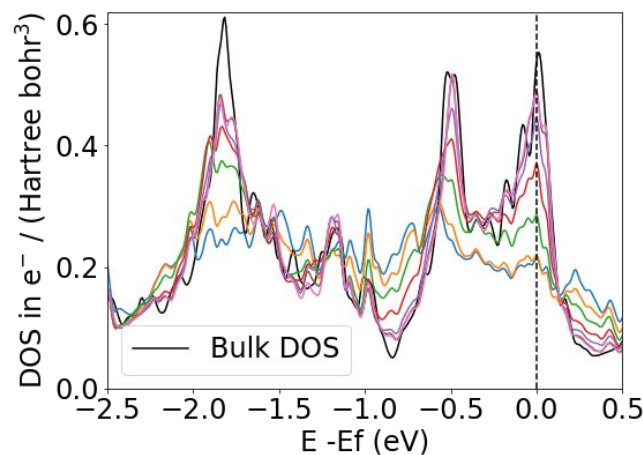
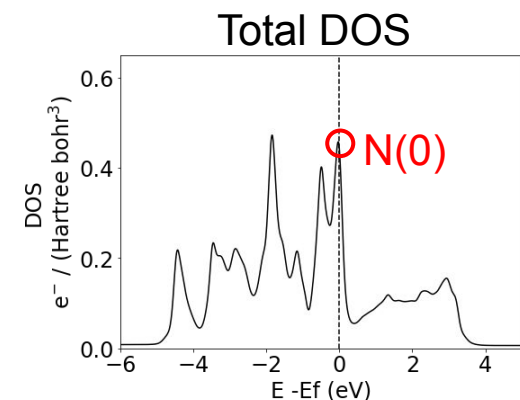




Local DOS vs distance from a “clean” GB

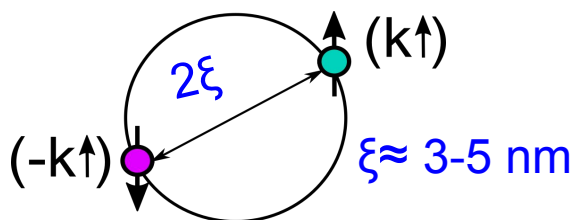
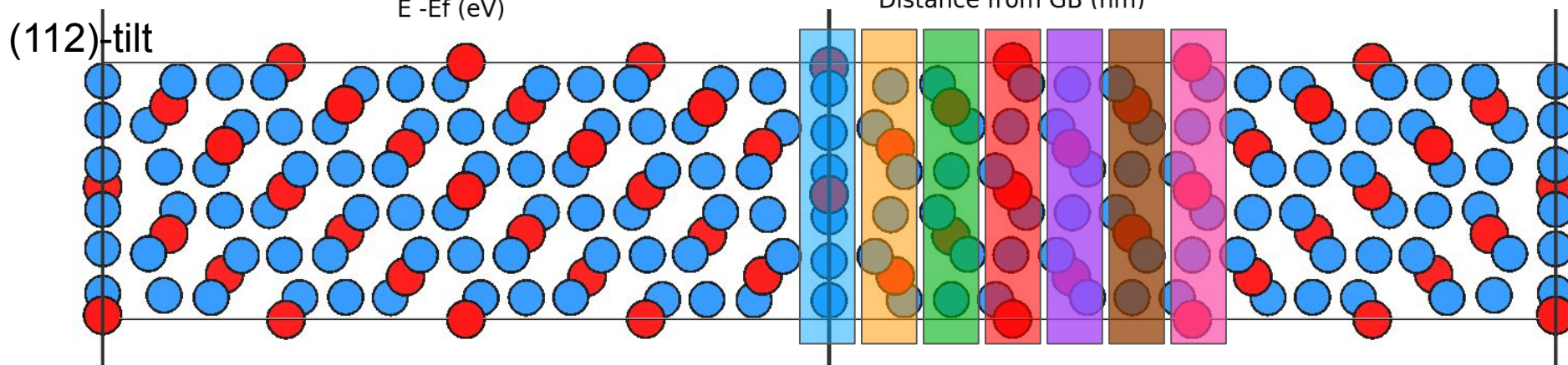
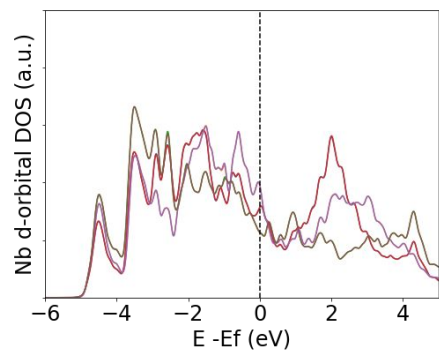


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$N(0)$ relaxes back to bulk in $\sim 1\text{-}1.5 \text{ nm}$

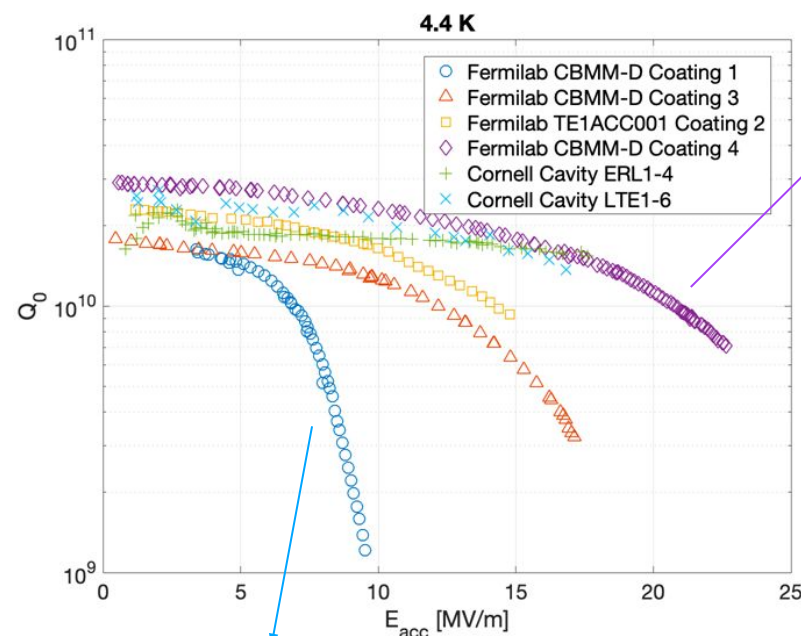
Nb d-orbital DOS in GB



Cannot directly translate **local** $N(0)$ into a **local** T_c since $\sim 1\text{-}1.5 \text{ nm} < \sim 3\text{-}5 \text{ nm}$
(Superconductors should not be sensitive to defects $< \xi$)



Tin antisite defect binding energies to GBs

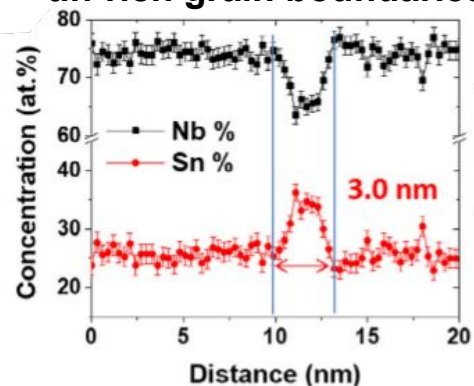


Stable Q in cavities with
clean grain boundaries

What causes tin to segregate towards GBs?

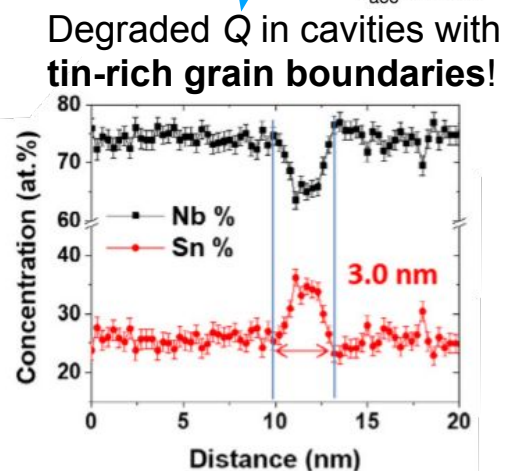
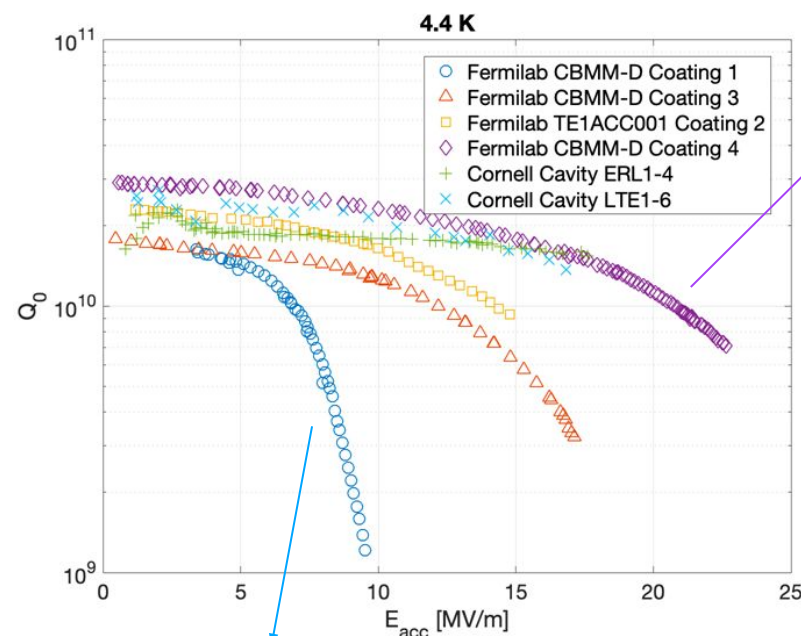
Tin antisite defect study

Degraded Q in cavities with
tin-rich grain boundaries!

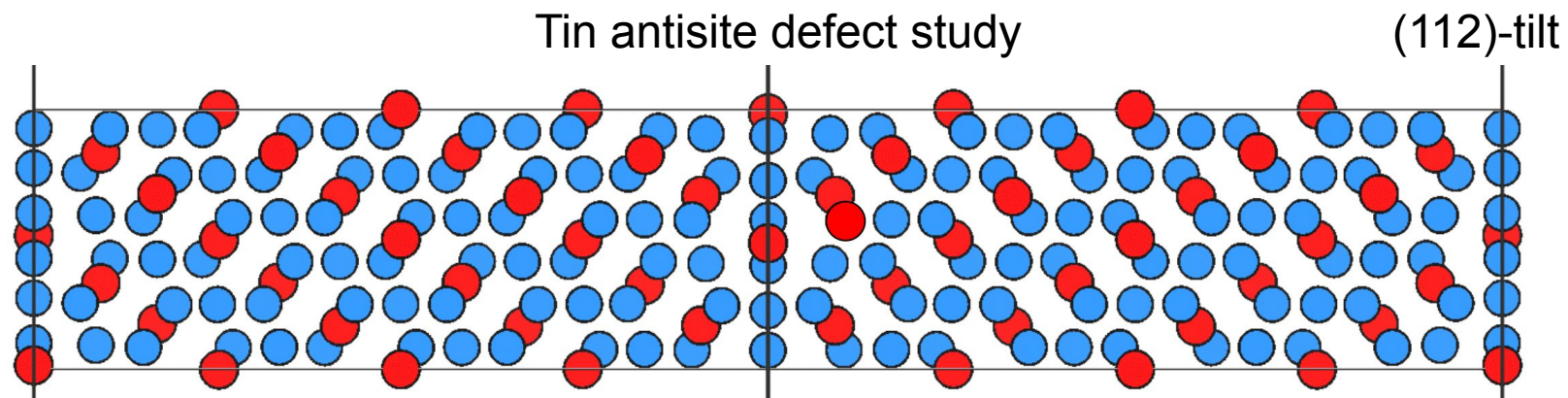




Tin antisite defect binding energies to GBs

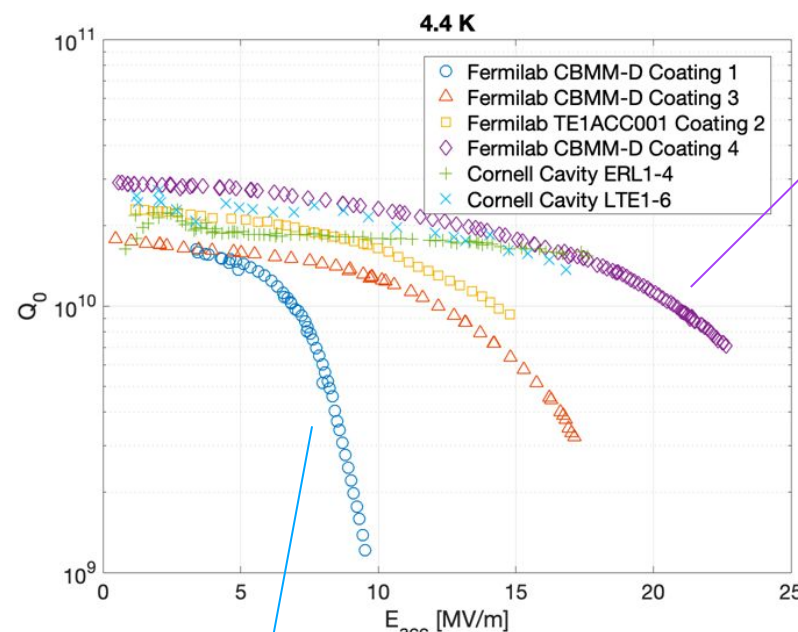


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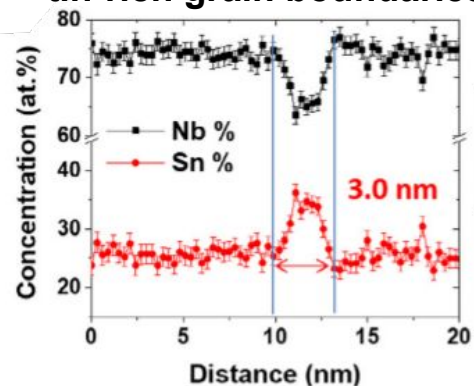




Tin antisite defect binding energies to GBs



Degraded Q in cavities with tin-rich grain boundaries!

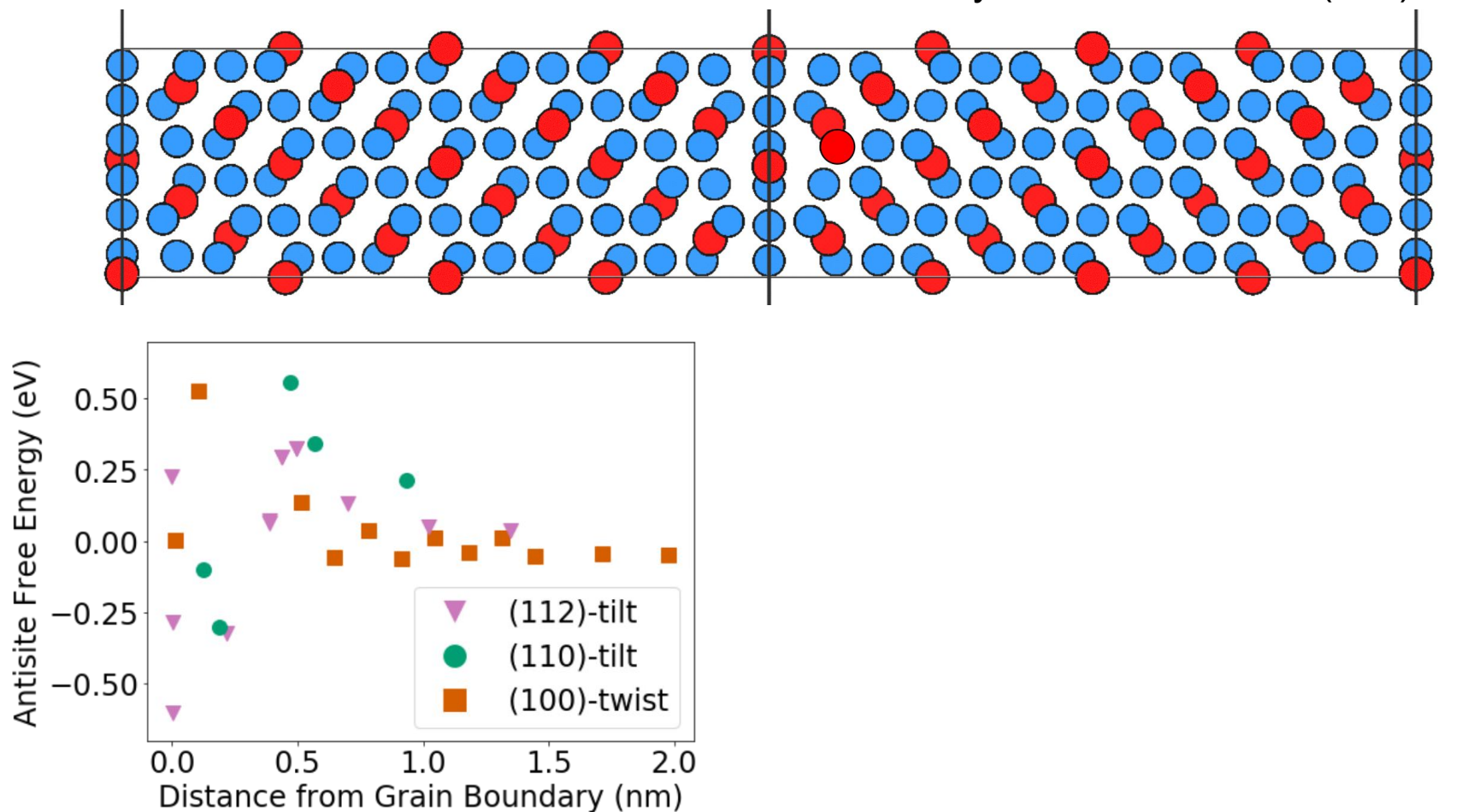


Stable Q in cavities with clean grain boundaries

What causes tin to segregate towards GBs?

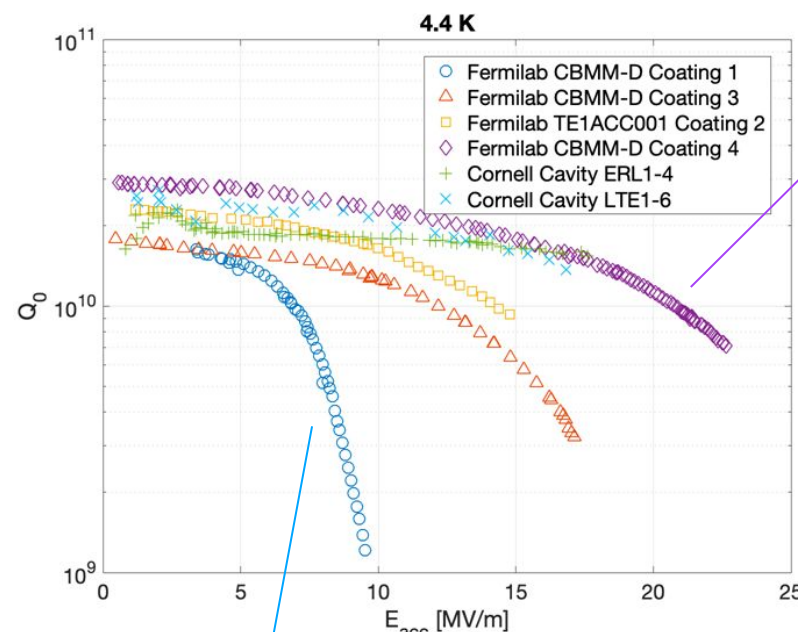
Tin antisite defect study

(112)-tilt

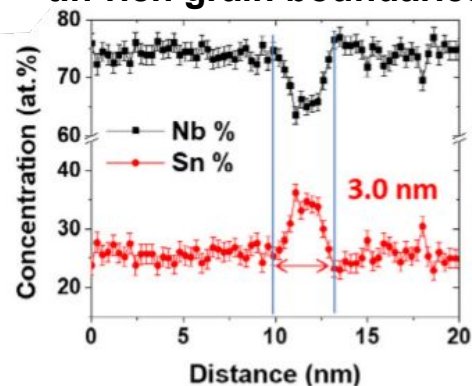




Tin antisite defect binding energies to GBs



Degraded Q in cavities with tin-rich grain boundaries!

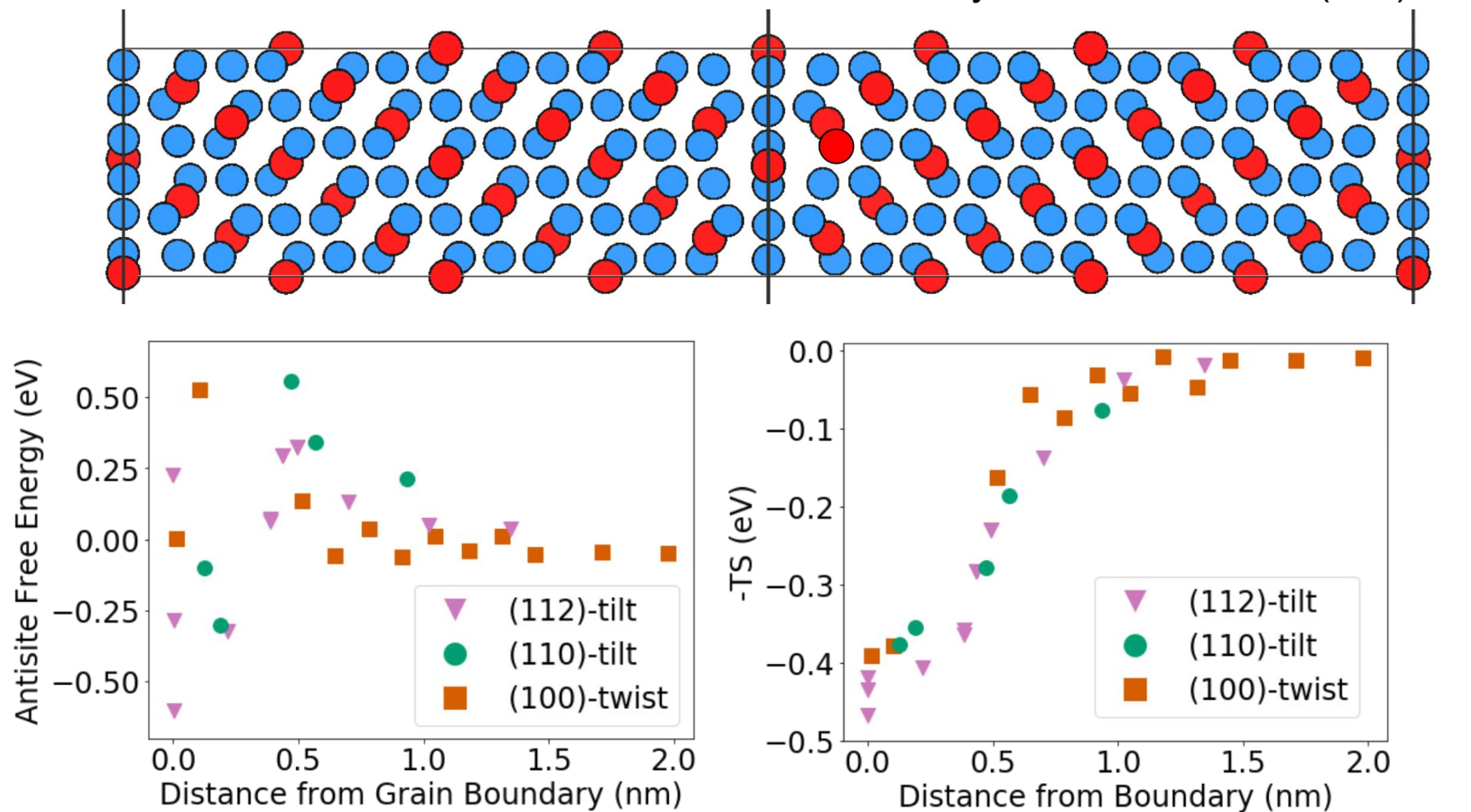


Stable Q in cavities with clean grain boundaries

What causes tin to segregate towards GBs?

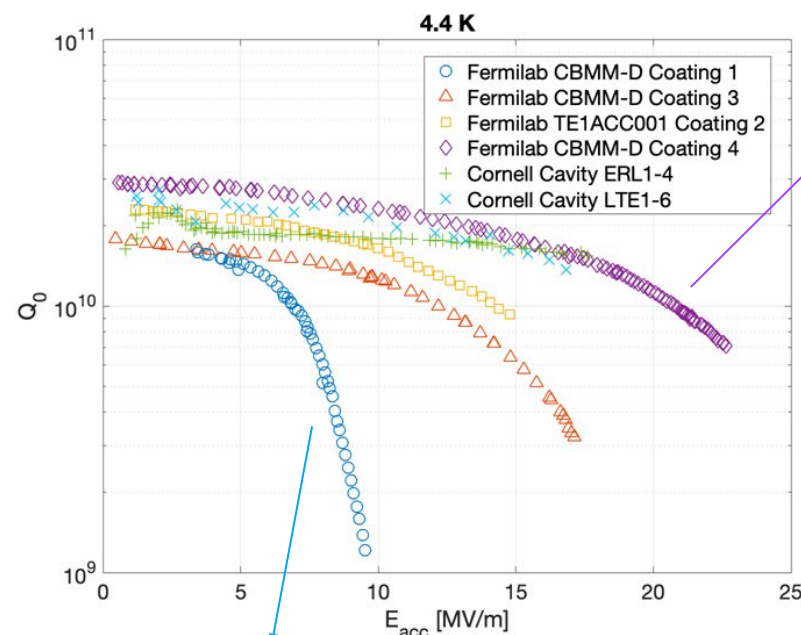
Tin antisite defect study

(112)-tilt

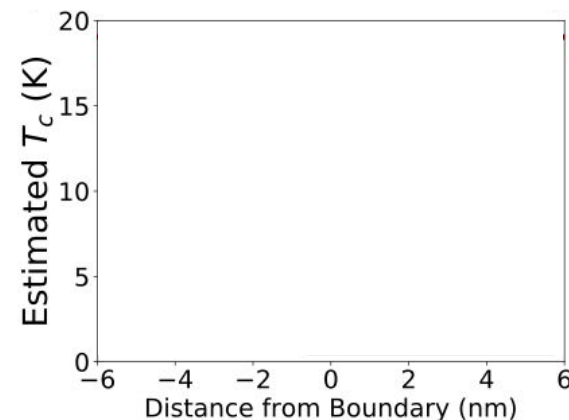
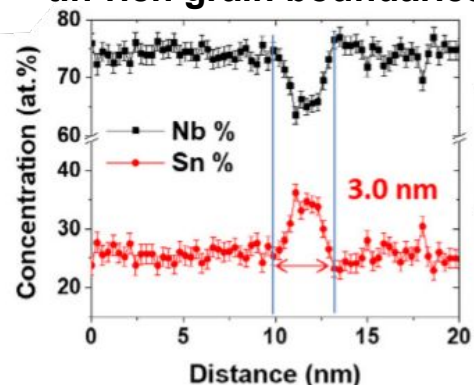




Model for local T_c with GB composition

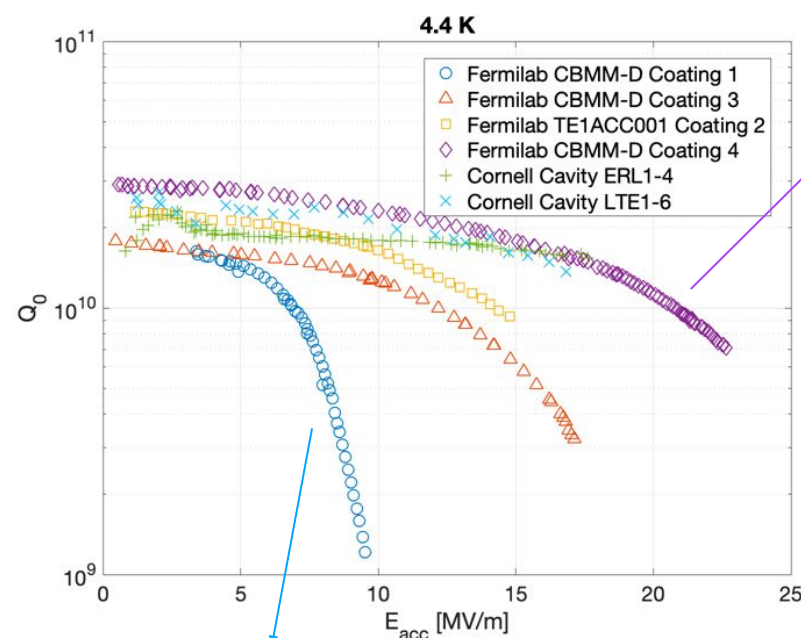


$$k_B T_c = 1.13 E_D \exp(-1/(N(0)V))$$



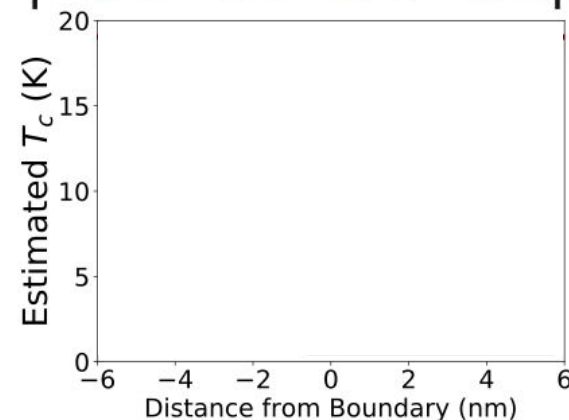
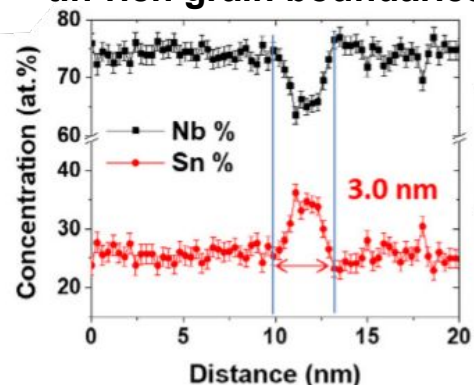
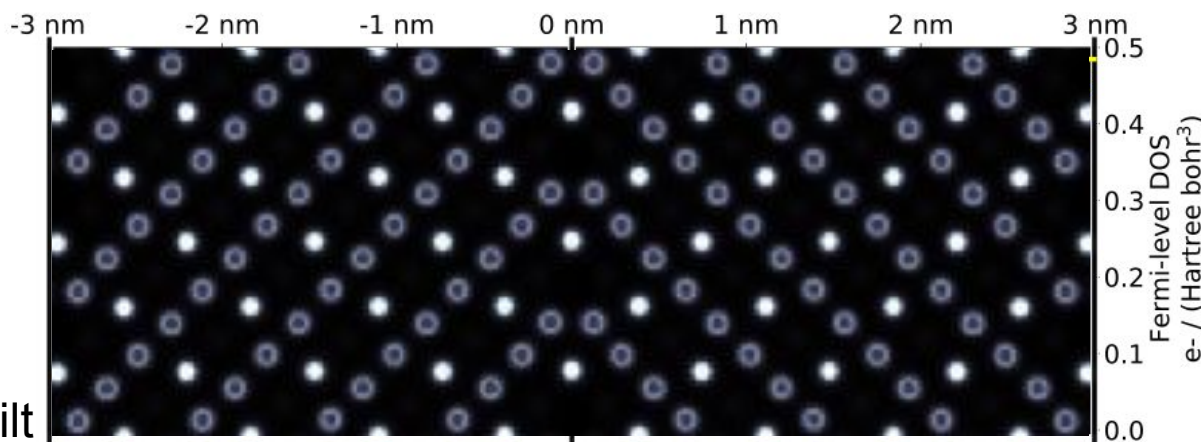


Model for local T_c with GB composition



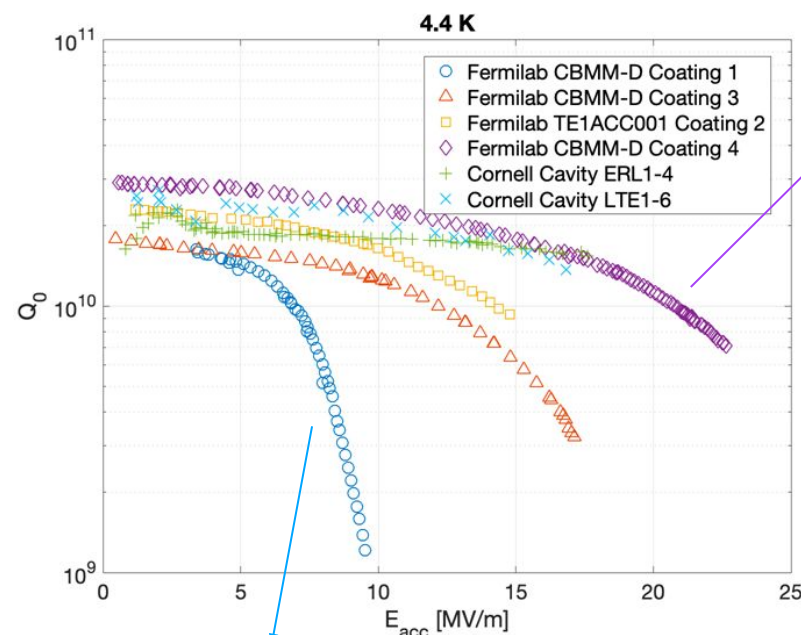
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(110)-tilt

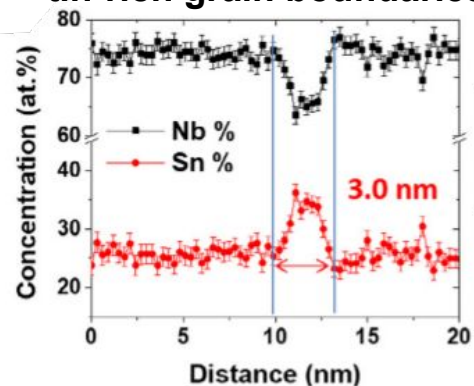




Model for local T_c with GB composition



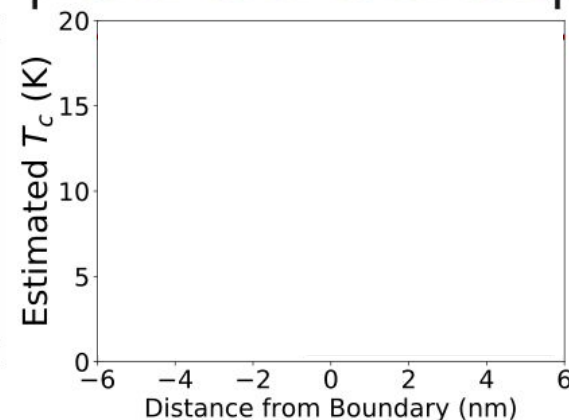
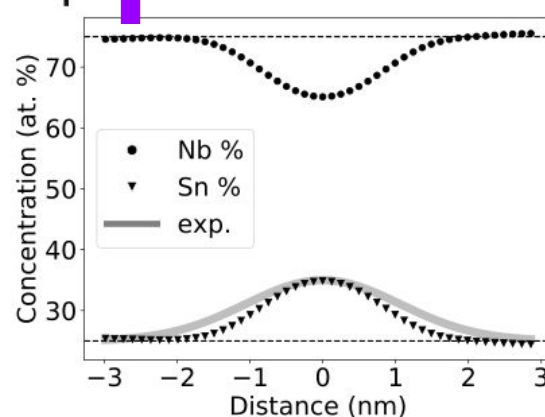
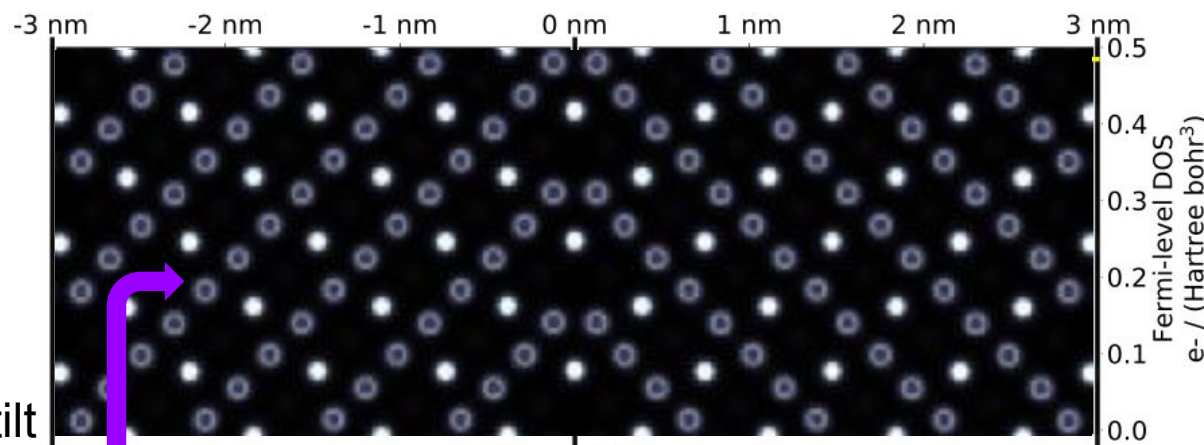
Degraded Q in cavities with **tin-rich grain boundaries!**



(110)-tilt

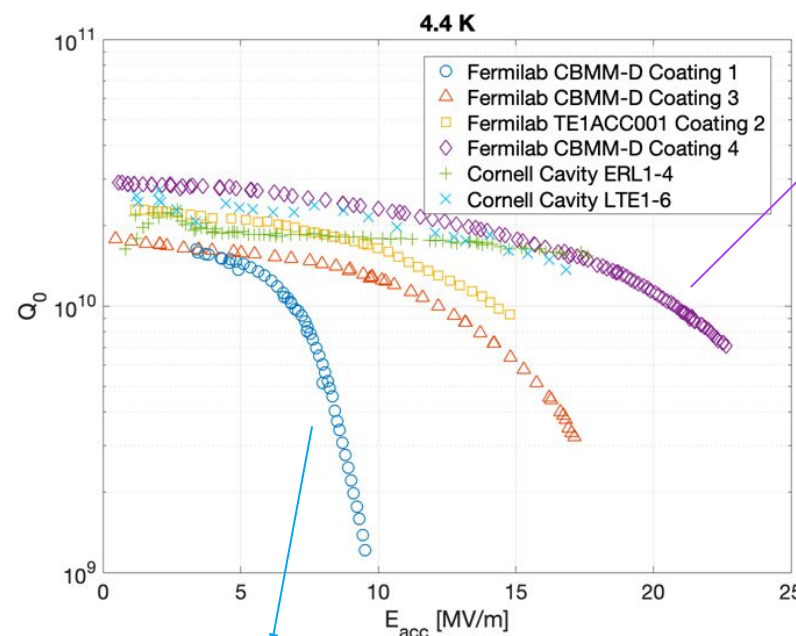
Stable Q in cavities with **clean grain boundaries**

$$k_B T_c = 1.13 E_D \exp(-1 / (N(0) V))$$

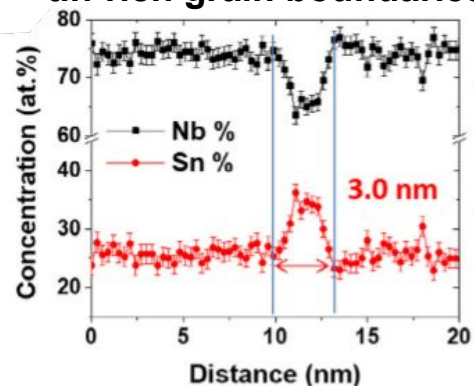




Model for local T_c with GB composition



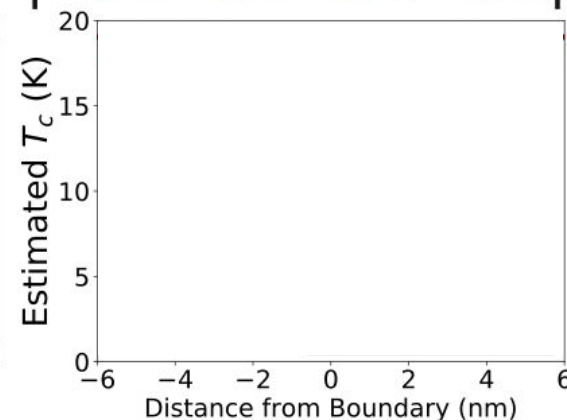
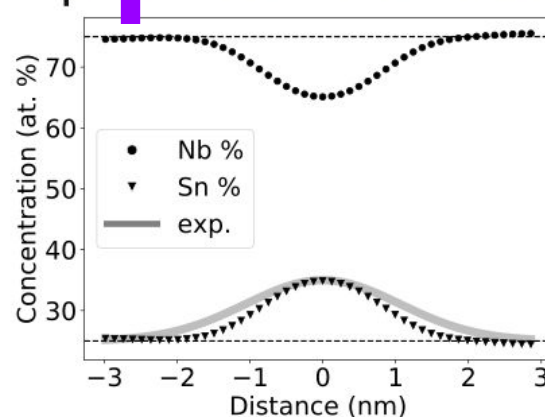
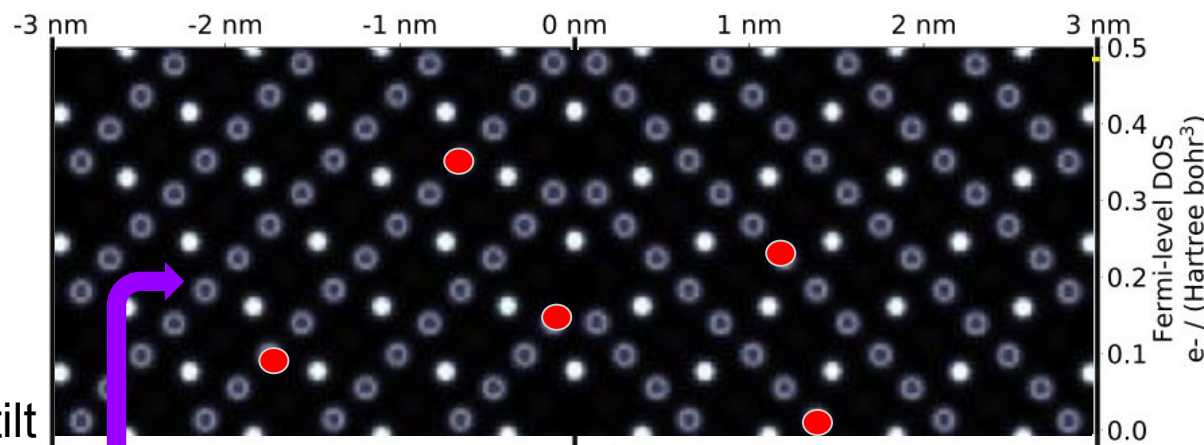
Degraded Q in cavities with **tin-rich grain boundaries!**



(110)-tilt

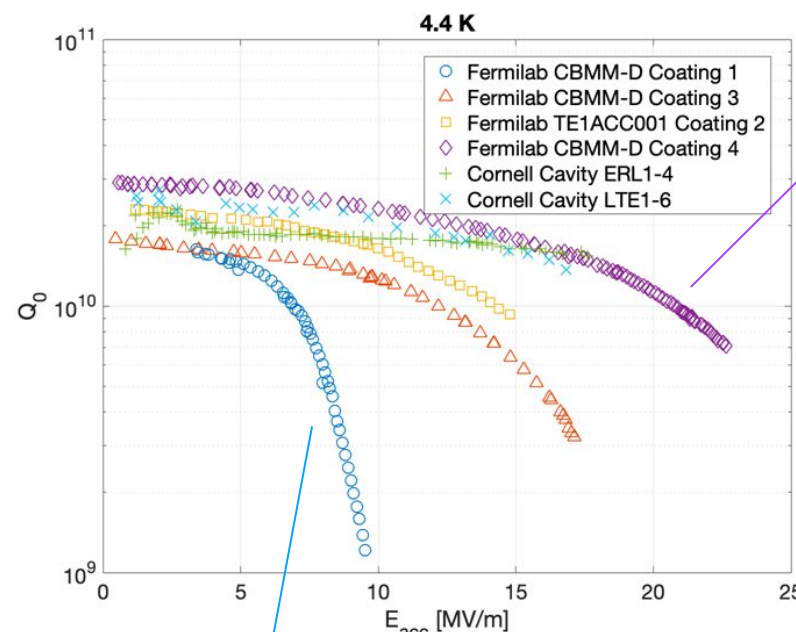
Stable Q in cavities with **clean grain boundaries**

$$k_B T_c = 1.13 E_D \exp(-1 / (N(0) V))$$



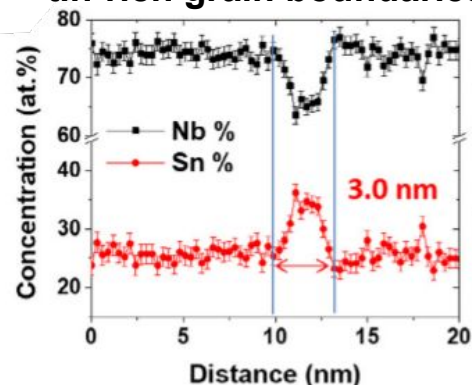


Model for local T_c with GB composition

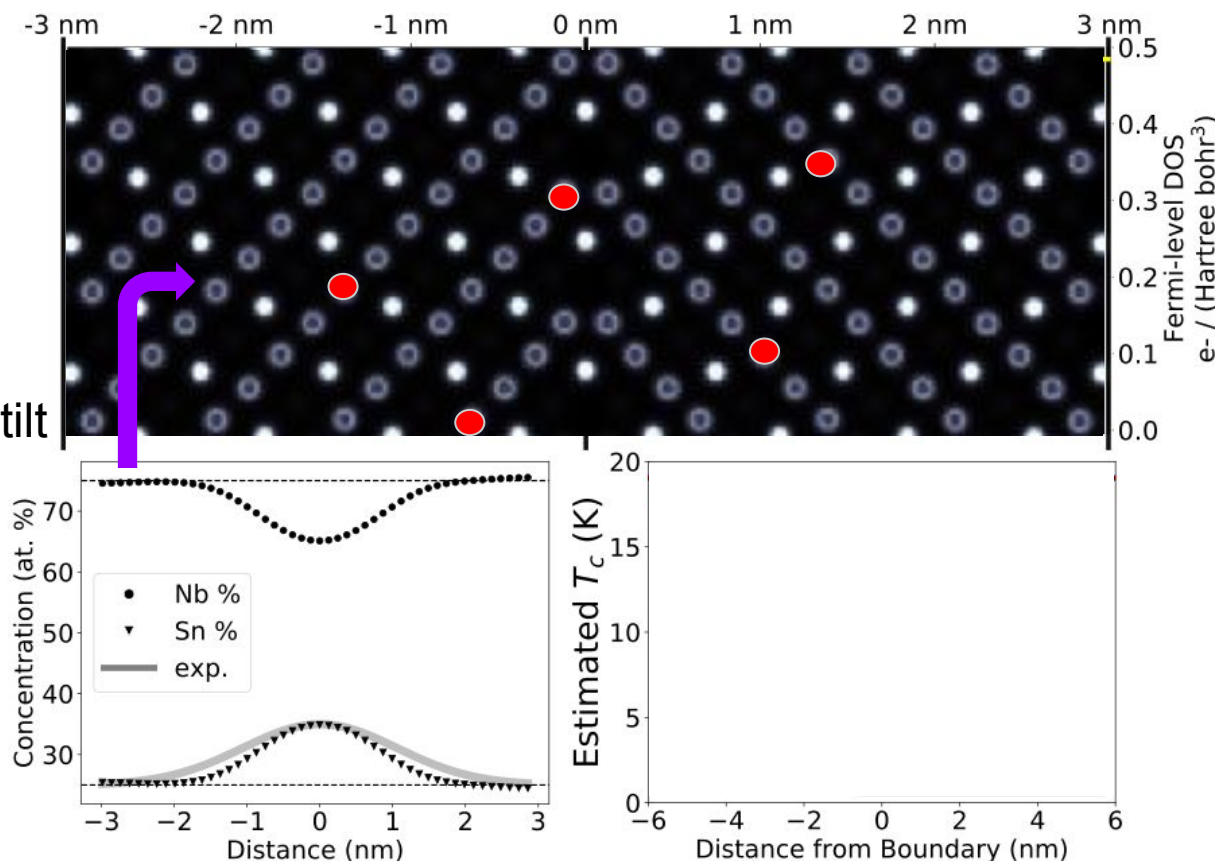


$$k_B T_c = 1.13 E_D \exp(-1/(N(0)V))$$

Degraded Q in cavities with tin-rich grain boundaries!

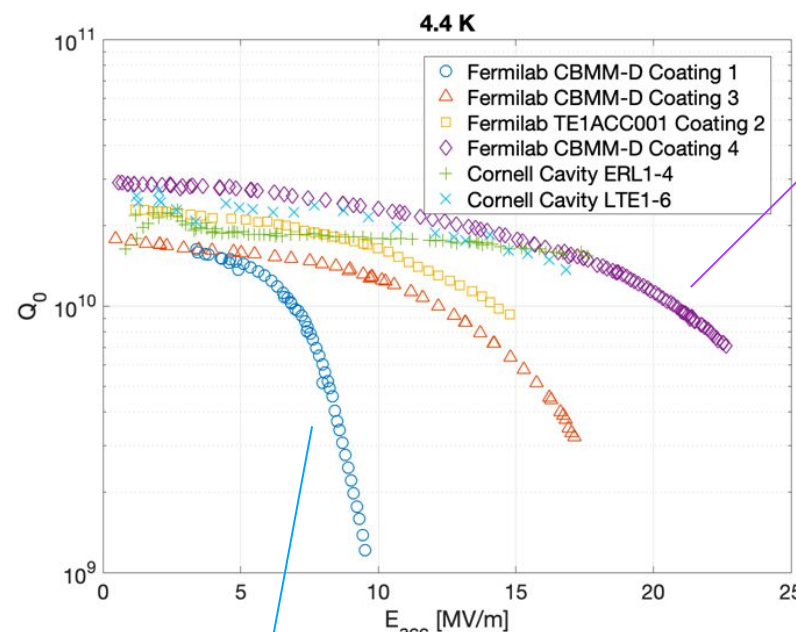


(110)-tilt



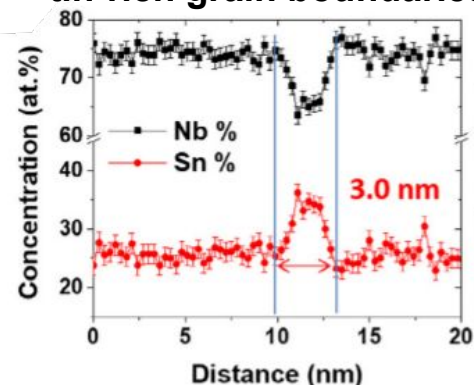
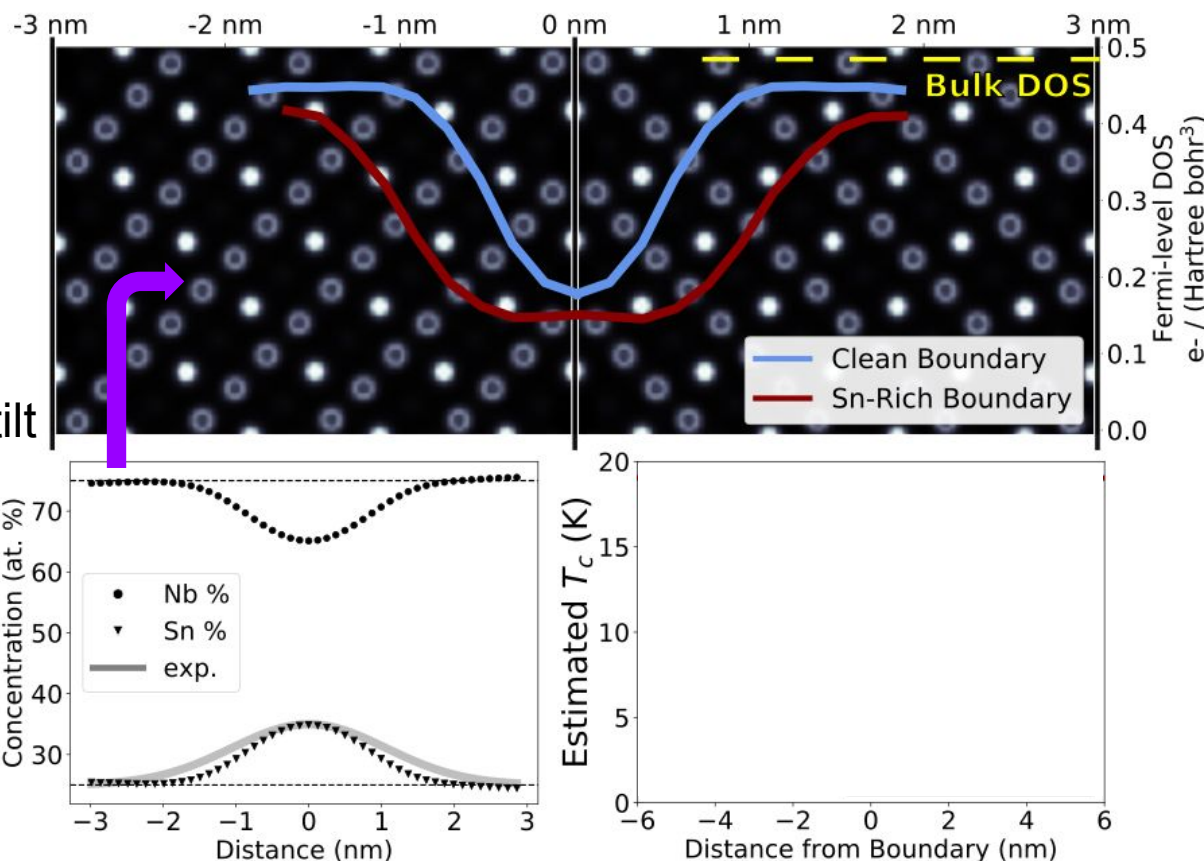


Model for local T_c with GB composition



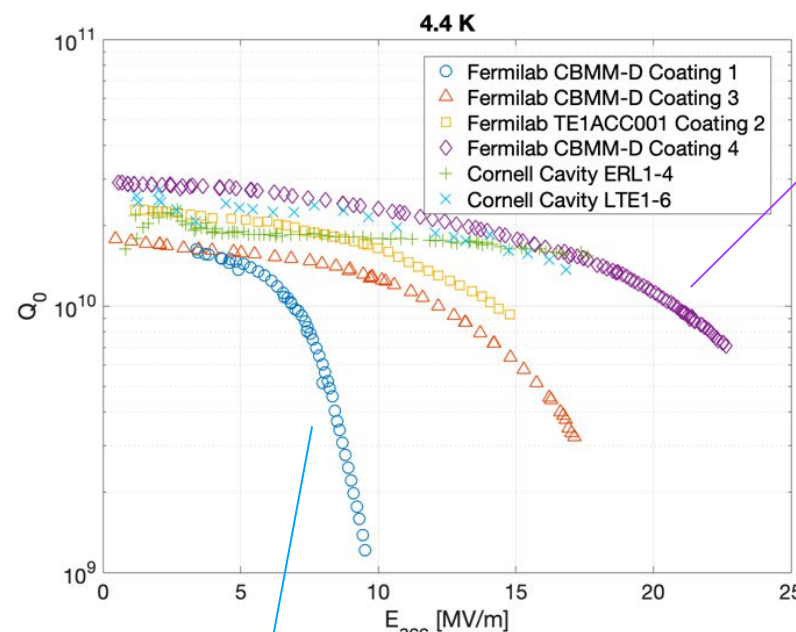
$$k_B T_c = 1.13 E_D \exp(-1/(N(0)V))$$

(110)-tilt

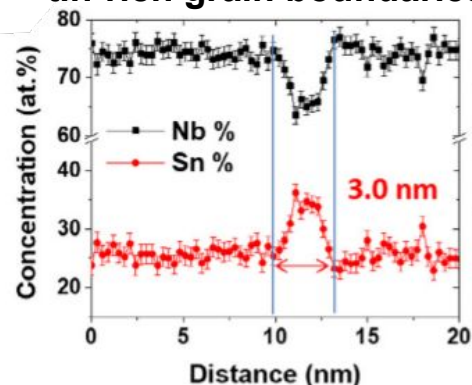




Model for local T_c with GB composition

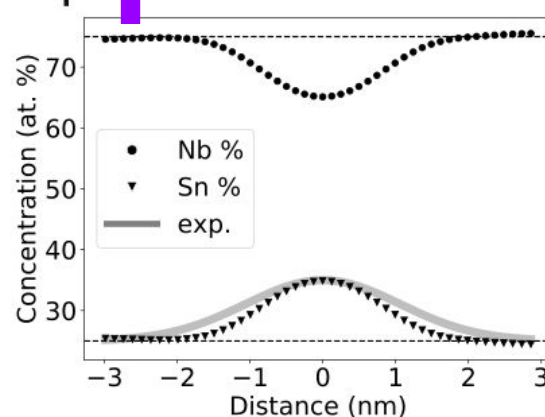
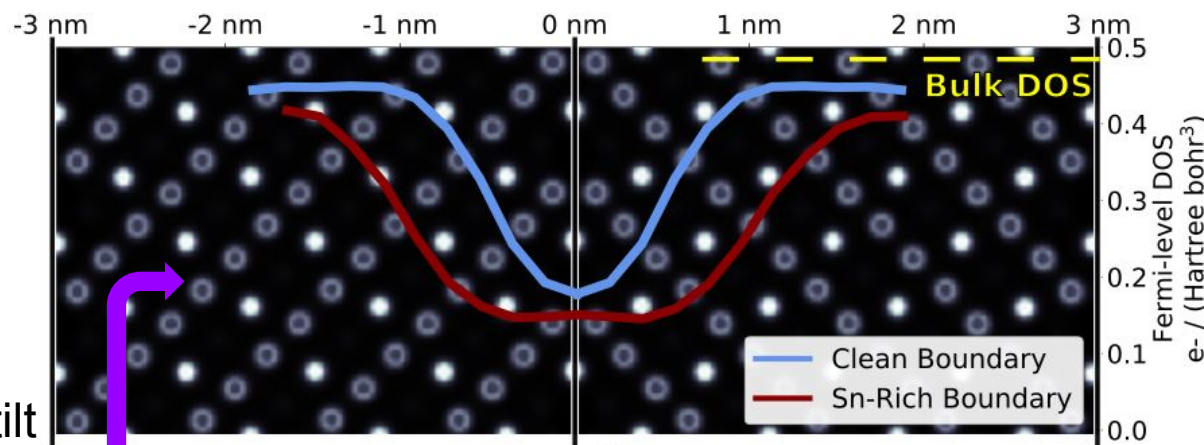


Degraded Q in cavities with **tin-rich grain boundaries!**

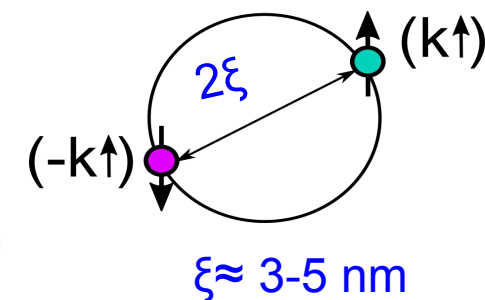


(110)-tilt

Stable Q in cavities with **clean grain boundaries**

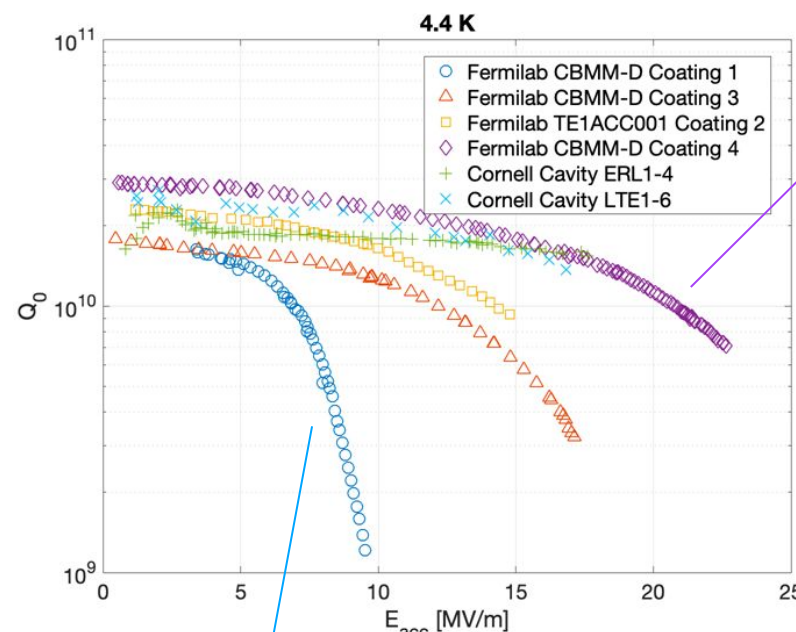


$$k_B T_c = 1.13 E_D \exp(-1 / (N(0) V))$$

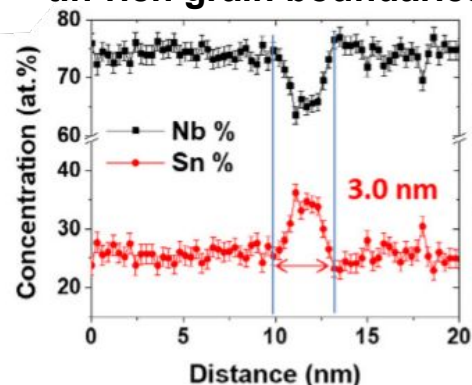




Model for local T_c with GB composition



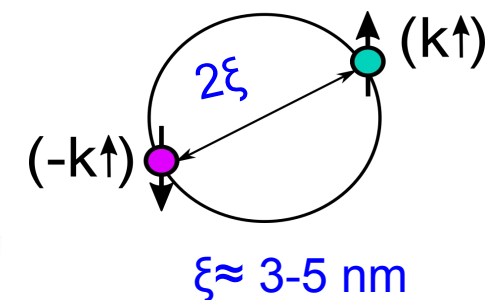
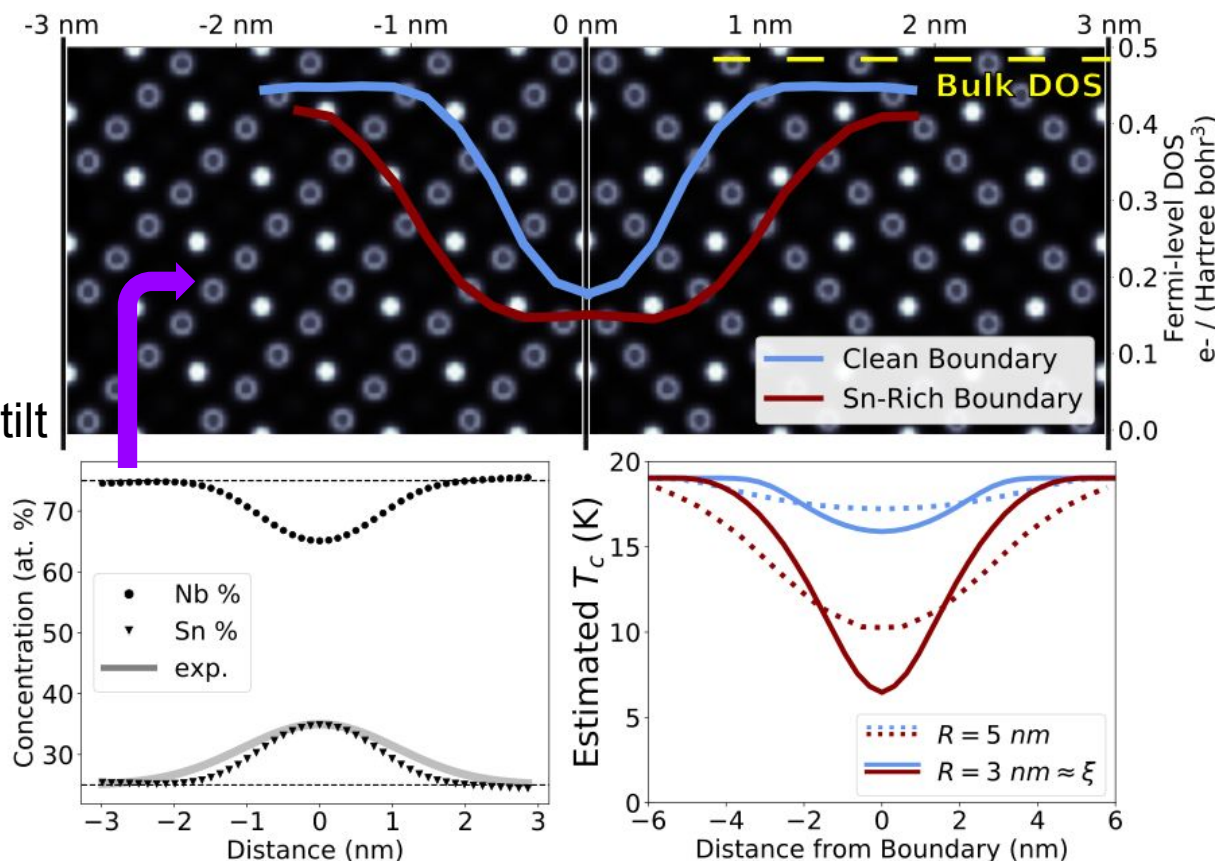
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Stable Q in cavities with **clean grain boundaries**

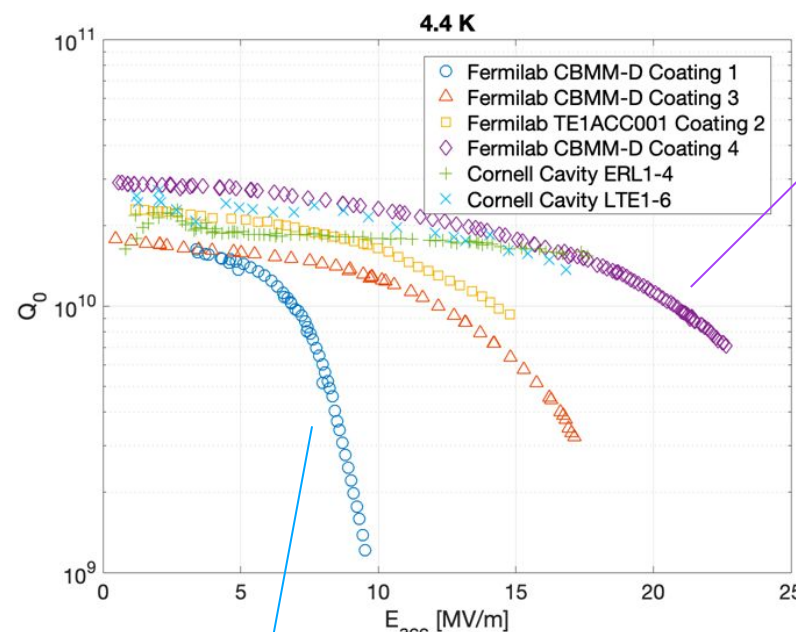
$$k_B T_c = 1.13 E_D \exp(-1 / (N(0) V))$$

(110)-tilt

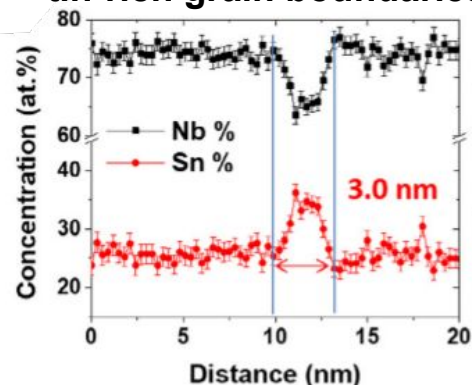
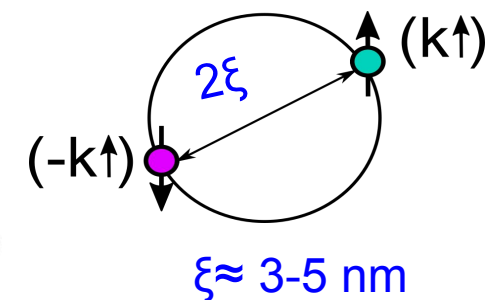
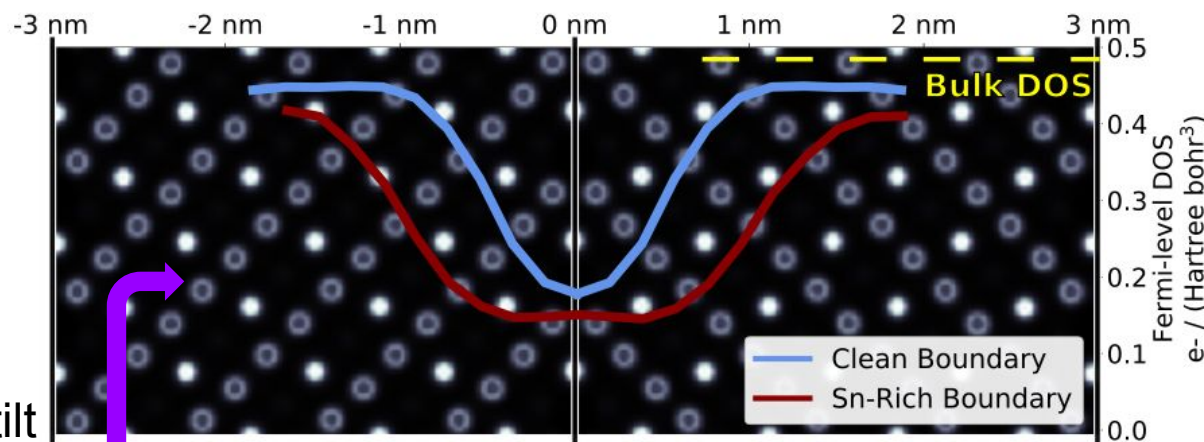




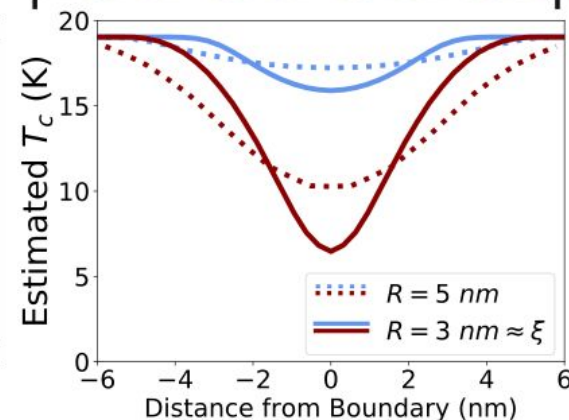
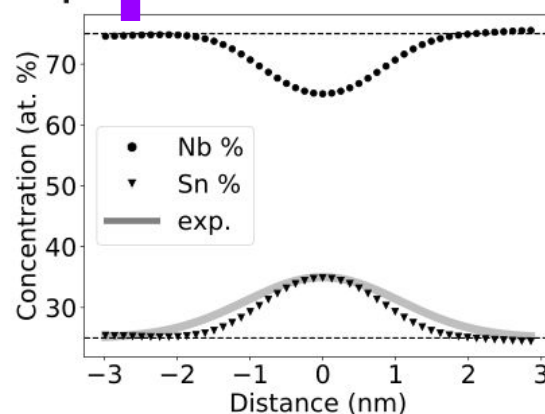
Model for local T_c with GB composition



$$k_B T_c = 1.13 E_D \exp(-1/(N(0)V))$$



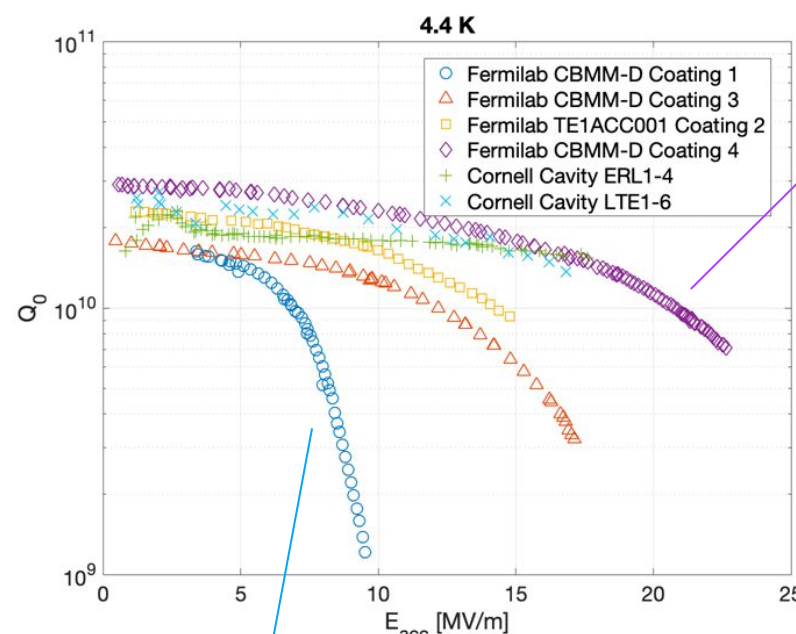
(110)-tilt



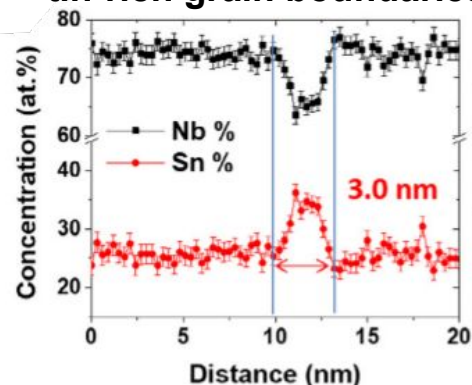
Clean GBs hardly affect T_c



Model for local T_c with GB composition



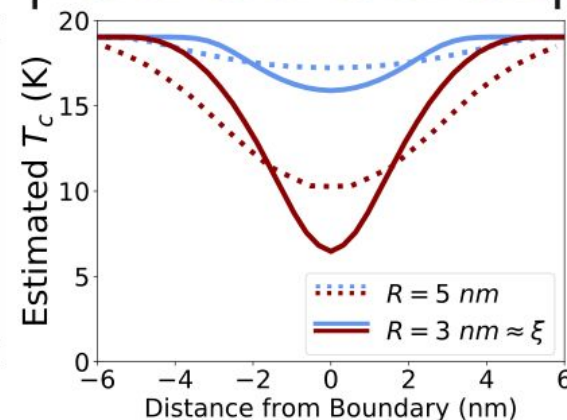
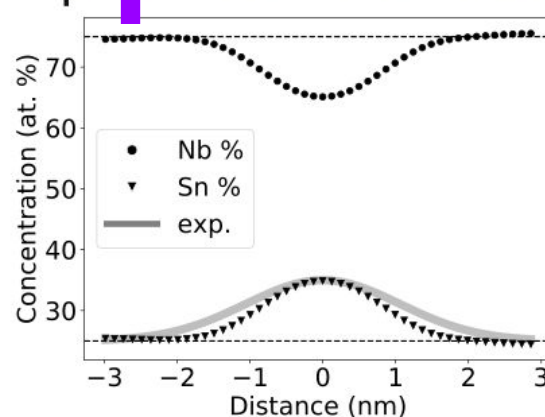
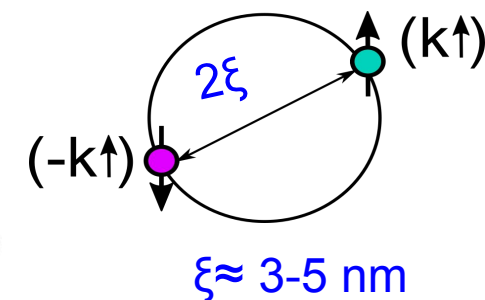
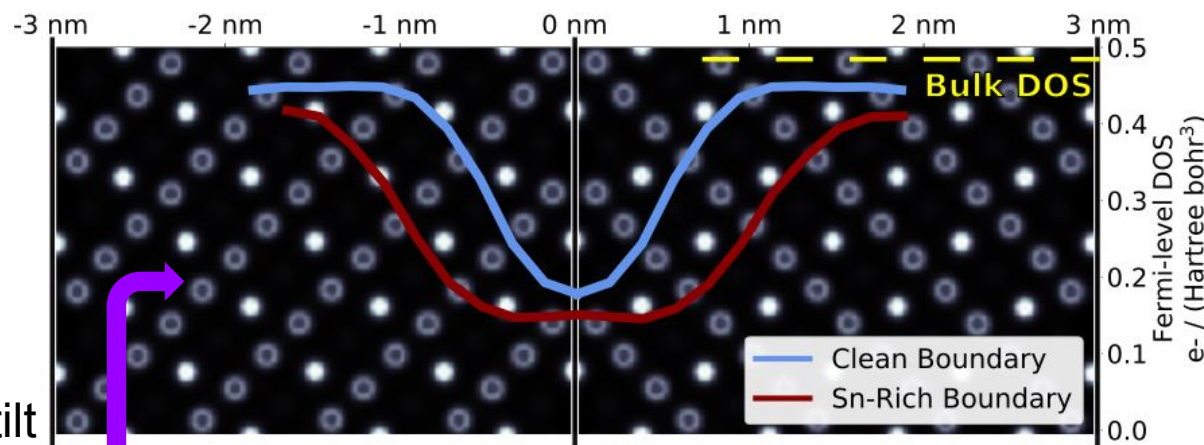
Degraded Q in cavities with tin-rich grain boundaries!



(110)-tilt

Stable Q in cavities with clean grain boundaries

$$k_B T_c = 1.13 E_D \exp(-1/(N(0)V))$$



Clean GBs hardly affect T_c

Tin-rich GBs have large impact on T_c → highly degraded superconducting properties!



Conclusions





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Too high of tin chemical potential during Nb_3Sn cavity growth will cause excess tin to diffuse towards GBs which should be avoided



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Speculations:

$\Delta N(0)$ heavily influences properties of Nb_3Sn → inexpensive DFT calculations can help predict defect segregation behavior



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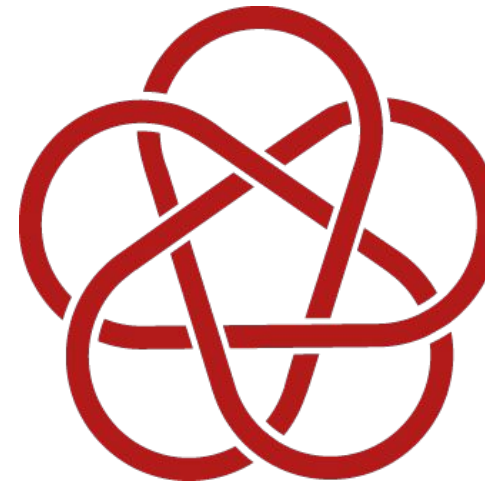
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More on this work:

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